



MSI  EUREKA | 通往无机材料的大门

**iGroup · 上海**

2019

# 提纲

## 一、MSI公司及产品介绍

## 二、MSI数据库客户端使用介绍

# 公司总览

MSI EUREKA

MSIT  
MATERIALS SCIENCE  
INTERNATIONAL TEAM

MSI  
Science Simplified

## 产品——无机材料数据库

- ▶ 收集所有发行及出版的数据资料
- ▶ 数据整合：去除受争议及无用的数据
- ▶ 主题：相图、相反应、热力学

## 团队

- ▶ 1984年由来自全球的材料科学专家共同组成
- ▶ 收集&评估数据, 补充缺失数据, 测试新数据

## 公司

- ▶ 提供全球化的材料科学类服务
- ▶ 管理并推广产品

# 起源&发展

## 源于德国马普学会

- ▶ 1984年,一个材料科学家团队(MSIT)在德国马普学院进行金属研究;
- ▶ 1989 年MSI GmbH为MSIT提供研发条件;
- ▶ 今天, MSI与MSIT共同形成了世界上最大的材料领域的全球化研究网络。



MAX-PLANCK-GESELLSCHAFT

## Dr Gunter Effenberg

- ▶ MSI的CEO和创立者; APDIC (国际合金相图委员会) 的创始成员之一; 担任过APDIC的主席; 美国材料信息学会 (ASM International)、 European Awards Committee的主席; 德国材料学会的成员。



MSIT  
MATERIALS SCIENCE  
INTERNATIONAL TEAM

## 一个全球化的协作团队

35年历史~250名材料科学家远程协作

- ▶ 监测所有相关出版物
- ▶ 评估二元&三元系数据
- ▶ 每年召开一次研讨会
- ▶ 开展合作项目

MSI EUREKA

*by Scientists for Scientists!*

### MSIT分布地点

GB Leeds; Sheffield; Manchester; Birmingham; Surrey  
 DE Stuttgart; Clausthal; Aachen; Jülich; Freiberg  
 NL Eindhoven  
 FR Lille; Montpellier; Rennes; Paris; Grenoble; Lyon  
 BE Leuven  
 AT Vienna  
 IT Genova  
 GR Volos  
 UA Kiev (Acad. Sci.); L'viv (Univ.); Chernivtsi; Kramatorsk  
 RU Moscow (Acad. Sci.); State Univ.  
 CN Changsha / Hunan; Central South Univ.; Beijing  
 JP Tokyo (IT); Kyoto, Sendai  
 Malaysia Sains Univ. Tronoh  
 USA Cincinnati; Raleigh; Gainesville; Evanston; Gaithersburg  
 BR Campinas; Lorena; Sao Paulo, IPT; PUC Rio  
 South Africa Witwatersrand  
 India Chennai, Bhabha Atom. Center (Mumbai)

# MSI & MSIT出版物

## ► “Ternary Alloys”(三元合金)丛书，全18卷

对材料体系的严格评估；铝、银、砷、锂、镁三元系相图；曾与Wiley出版社共同出版，后由MSI独立出版

## ► Landolt-Börnstein(简称LB工具书)子系列 “Ternary Alloys Phase Diagrams” (三元合金相图)，全17卷

对材料体系的严格评估；与Springer出版社共同出版

## ► “Red Book” 系列:"Phase Diagrams of Metallic Systems"丛书，全18卷

原由前苏联科学院出版，出版全球相图相关文献的标准摘要，报道前一年的科学进展。年鉴，因封面总是红色而得名；现与俄罗斯国家科学技术信息研究所(VINITI)共同出版，现只有网络版

## ► “Metal-Boron-Carbide”

作者：Peter Rogl, 由MSI编辑，并与美国材料信息学会(ASM)共同出版

## ► “Pressure Dependent Binary Phase Diagrams”

作者：Yuri Lewinski, 由MSI编辑，并与美国材料信息学会(ASM)共同出版





# 主题: Phase Transformations (相的转化)

- 材料性质会因其相变而受影响
- 相会因温度、压力或材料成分的改变而变化

## 因温度变化



温度升高将固相（雪）转化为液相

## 因压力而变化



冰鞋产生的压力将固体(冰)转化为液体。  
滑冰者实际上实在一层水膜上滑行。

## 因成份而变化



在雪地上撒盐(冰+氯化钠)以降低  
熔点。冰转化为液体。

- 科学家从**相图**中了解这些相变
- 就在 MSI EUREKA

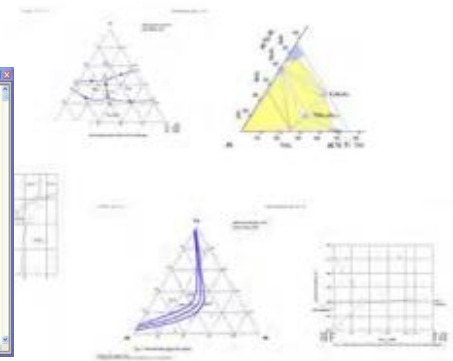
## MSI Eureka——数值/工具型数据库

MSI Eureka持续关注无机材料的全球出版物，并汇编1894年以来一元、二元、三元及多元系统的相图及其相关数据等。提供：

- ▶ 评估已发表数据
- ▶ 整合分散的数据
- ▶ 相图、微观结构图、性质、热力学数据
- ▶ 帮助了解材料

Table 2: Invariant Equilibria

T (°C)	Type	Phase	Composition (at %)
Reaction			Al Cu Mg
665	$L + \eta_2 + \eta_{13} = V$	$P_1$	45.9 42.9 12.1
538	$L + \eta_{13} = Q$	$P_2$	53.6 6.7 39.7
994-1022	$L + \beta_2 = \beta + \eta_2$	$U_1$	
818	$L + \eta_2 = \eta_1 + \beta$	$U_2$	25.5 58.7 15.8
775	$L + \eta_2 = \eta_1 + \lambda$	$U_3$	36.3 56.8 14.9
724	$L + \eta_2 = \eta_1 + \eta_{13}$	$U_4$	39.5 48.2 14.3
595		$U_5$	53.5 38.1 8.4





# MSI Eureka内容简介

## 数据汇编

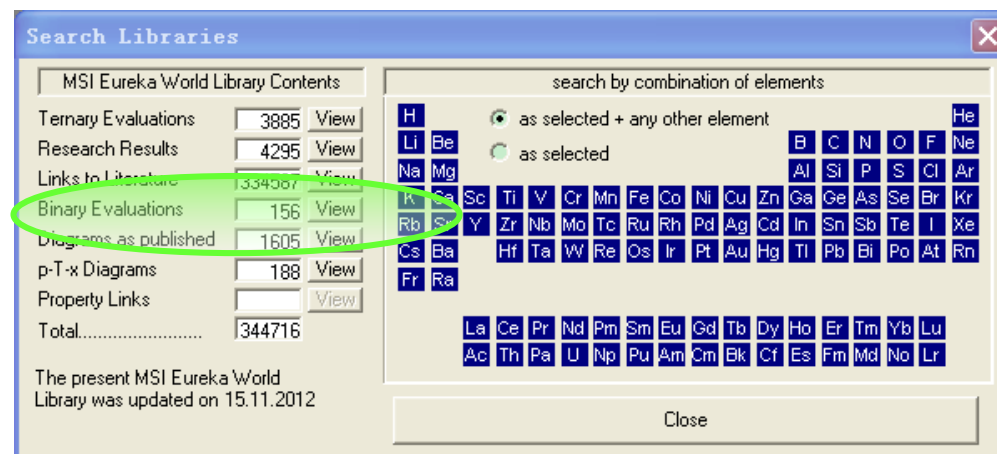
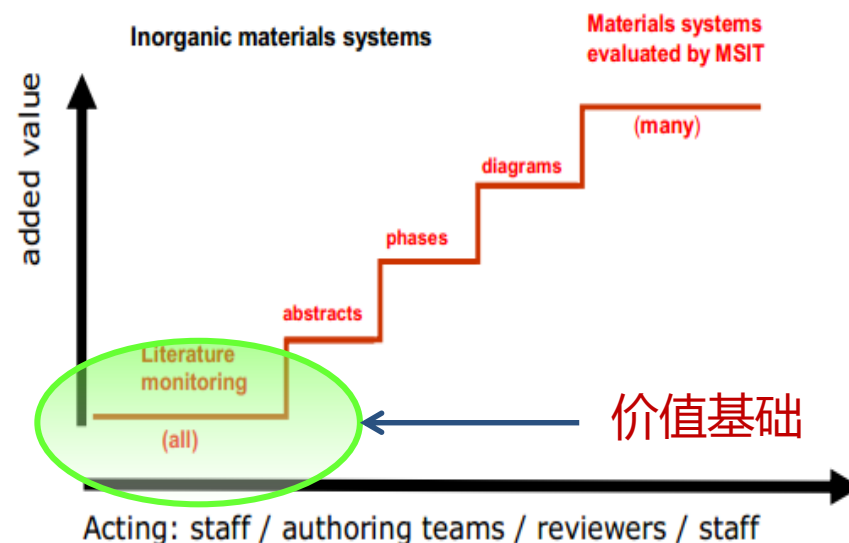
- 文献---由MSIT编辑
  - 收录的书目数据~448,000条
  - 所有出版过的无机材料体系~71,000个(不含无机盐)
- 评估报告
  - 三元系~4000个, 二元系245个, 单元体系188个

## 参考相图

- 1900张相图, 含1600个二元及三元体系

## 特色

- 同行评议: 经材料工作者们多次验证
- 材料齐全, 参考文献范围广泛
- 界面简洁方便, 搜索结果高度匹配
- 个性化服务功能



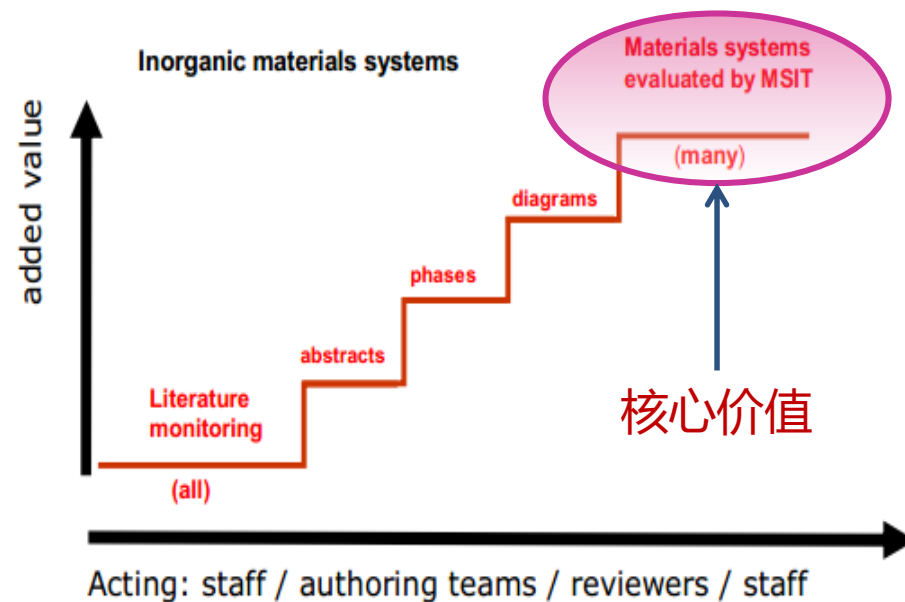
# MSI Eureka内容简介

数据评估---严格测试过的材料体系

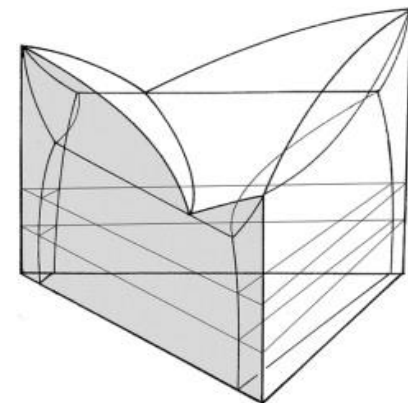
- ▶ 三元系: ~4000个
- ▶ 二元系: 245个
- ▶ p-T-x: 188个压力/温度数据

## 特色

- ▶ 将所有出版物及描述汇集一体
- ▶ 评估 & 证实实验结果
- ▶ 检查数据的一致性并对其进行解释
- ▶ 指出并解决争议或矛盾数据
- ▶ 囊括非公开发表数据
- ▶ 将数据整合到相图中



Search Libraries		
MSI Eureka World Library Contents		
Ternary Evaluations	3885	View
Research Results	4295	View
Links to Literature	334587	View
Binary Evaluations	156	View
Diagrams as published	1005	View
p-T-x Diagrams	188	View
Property Links		View
Total.....	344716	
The present MSI Eureka World Library was updated on 15.11.2012		



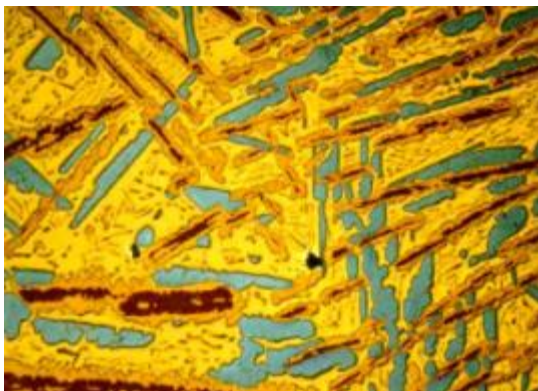
# 材料范围

## 材料种类

- ▶ 合金: 钢铁, 铜合金, 磁铁, 电子材料...
- ▶ 非金属: 陶瓷, 半导体, 传感器...
- ▶ 复合材料: 金属陶瓷...

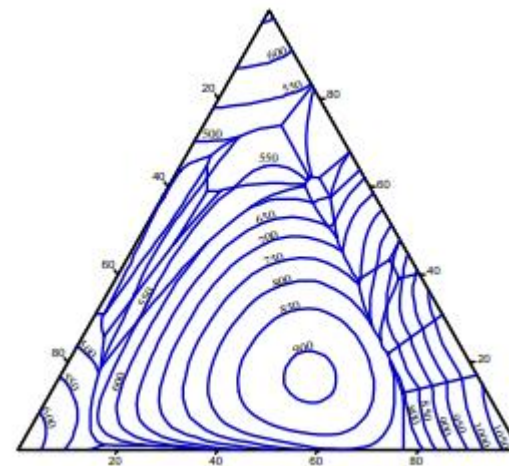
## 固相范围内的相及相反应

### 微观物相图



多重物相, Ag-Cu-Sn三元系

MSI EUREKA



等温相图, 材料研究者的地图

# 材料范围

铝合金	Al*	铜合金	Cu*
钢及铁合金	Fe*	稀土金属	Sc*, Y*, La*, Lanthanoids*
耐热合金	Ni*, Co*	硬质合金	W-C*, W-C-Co*, W-C-Cr*, W-C-Ni*, W-C-Nb*
轻金属合金	Al*, Ti*, Mg*, Be*, Li*		
硼化物陶瓷 碳氮合金	B*, C*, N*	高熔点金属 及其合金	Cr*, V*, W*, Nb*, Hf*, Zr*, Ta*, Mo*, Re*
锂离子电池	Li*, P*, Co*, Mn*, Ni*, Cu-Sn*, Ti-Si*, C-Si*, Mo-S*	贵重金属	Ag*, Au*, Pd*, Pt*, Rh*, Ru*, Ir*, Os*
焊接材料	Ag*, Au*, Bi*, Cu*, In*, Sb*, Sn*, Zn*, Pb*	半导体材料	Si*, Ge*, As*, N*, Ga*, Cd*, Te*, Se*, S*, Sb*, P*, In*, Cl*, I*, Br*, Zn-O*, Ti-O*, Cu-O*, U-O*, Bi-O*, La-O*, Fe-O*, Ni-O*, Eu-O*
核材料	Actinoids*, Cs*, O*, Zr*, Fr*, Ra*, Po*, At*, Rn*, Sr*, Xe*, H*, Ac*, Rf*, Sg*		

\*: 代表与其它元素组成的体系, 如Al\*代表Al与其它元素组成的体系

# 目标客户

- ▶ 科学家&工程师：学术界&企业界；科研专家；教育工作者&大学生
- ▶ 数据管理者：图书管理员；资讯专员
- ▶ 科学管理：项目经理；工程管理人员

## 学科领域

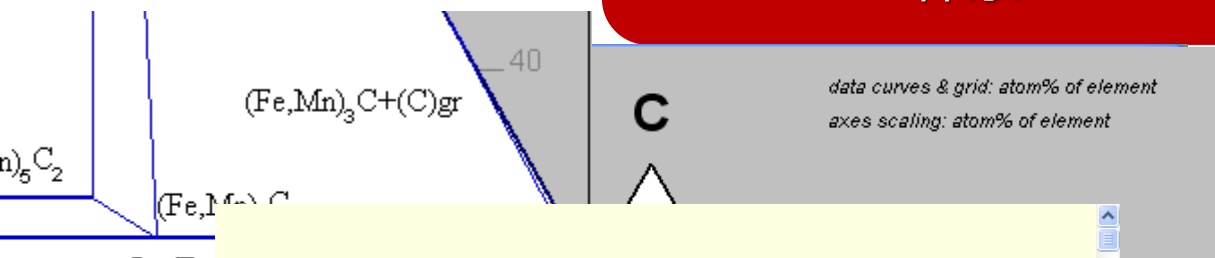
- ▶ 化学
- ▶ 物理
- ▶ 工程
- ▶ 材料学，包括材料设计
- ▶ 晶体学，晶体生长等
- ▶ 热力学
- ▶ 合金开发及设计
- ▶ 其他
- ▶ 为多种工业提供基本信息，包括自动化、航空航天，重工业，制造业，能源循环...
- ▶ 功能材料，传感器...
- ▶ 结构材料
- ▶ 金属，陶瓷，陶瓷合金
- ▶ 其它





## 使用案例

高铁铁轨需要高硬度、韧性、耐磨性的钢材料，如 Fe-C-Mn 体系+V+Cr



## References

## 1909Goe

Goerens, P., "About the Influence of Alloying Substances on Phase Diagram of Alloys of Iron and Carbon" (in German), Metallurgie, 6, 537-550 (1909) (Phase Relations, Experimental, 16)

## 1909Wue

Wuest, F., "Investigation of Influence of Manganese on the Phase Diagram Iron-Carbon" (in German), Metallurgie, 6, 3-14 (1909) (Phase Relations, Phase Diagram, Morphology, Experimental, 54)

## 1910Arn

Arnold, J.O., Read, A.A., "The Chemical and Mechanical Relations of Iron, Manganese, and Carbon", J. Iron Steel Inst., 81, 169-184 (1910) (Phase Relations, Mechan. Prop., Experimental, 12)

## 1927Had

Hadfield, R.A., Met, D., "Iron-Manganese Alloys Low in Carbon", Trans. Am. Inst. Min. Metall. Eng., 75, 440-442 (1927) (Mechan. Prop., Experimental, 1)

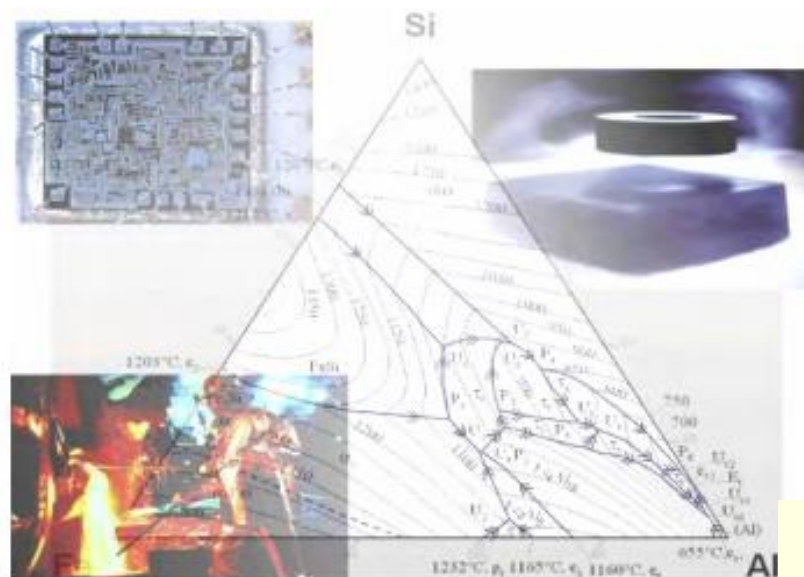
## 1932Bai

Bain, E.C., Davenport, E.S., Waring, W.S.N., Kearny, N.J., "The Equilibrium Diagram of Iron-Manganese-Carbon Alloys of Commercial Purity", Trans.

利用MSI设计新材料

1. 提供相关或相似体系材料的热力学参数；
2. 参考相似的研究方法，节约时间 & 资源

## 使用案例



制定材料的生产工艺或检测标准，如  
Fe-C-Mn

road maps to solutions

利用MSI查找材料参数

1. 提供最可靠的匹配参数，减少实验数量；
2. 检测所得材料的性能是否合格；
3. 为新项目的申请提供研究现状、可行性研究

Table 1: Recent Investigations of the C-Fe-Mn Phase Relations, Stru

Reference	Method / Experimental Technique	Temperature/Composition/ Phase Range Studied
[1989Dre]	C activity from equilibration of liquid with MnO containing slag and with CO containing gas phase	1400 to 1800°C, 70 and 80 mass% Mn, to 8 mass% C
[1990Ni1] [1990Ni2]	C solubility in melts	1400 and 1500°C, 0 to ~9 mass% Mn
[1990Put]	Metallography, XRD	827°C, 5 and 6 GPa, 70 to 100 at% C
[1993Wdt]	C partial enthalpy, enthalpy of formation, isoperibolic calorimetry	1627°C, 0 to 100% Mn, C up to saturation
[1994Prz]	C activities by equilibration with gas phase	900°C, to 1.55% C, 1.7% Mn
[1995Eno]	Mn activities by distribution between Fe and Ag	1463, 1500 and 1550°C, 0 to 0.45 Mn, C up to saturation

## 我们的优势

- ▶ **同行评议：**所有数据均经过材料工作者们的多次验证；
- ▶ **材料齐全，参考文献范围广泛：**  
71,000个无机材料体系，1894年至今的250种期刊及灰色文献
- ▶ **界面简洁方便，搜索结果高度匹配：**  
交互式元素周期表搜索方式
- ▶ **个性化服务功能：**动态、交互式图片，可放大、重叠显示和检索图片



# 访问方式

- ▶ 客户端下载:

<http://www.msiport.com/discover-msi-eureka/how-to-access-msi-eureka/eureka-viewer/download-installation/>

- ▶ 网页访问:

<https://search.msi-eureka.com/search>

## 客户赞誉

以前我们要在检索多个书目数据库上花费大量的时间。自从2000年开始使用MSIT Phase Diagram Center (MSIT相图数据库, EUREKA的前身), 做同样的工作只花几分钟的时间。节省的时间已是物有所值, 而且我们非常感激三元系相图数据的卓越品质。

*Prof. R. Schmid-Fetzer*  
德国克劳斯塔尔工业大学 (TU Clausthal)

有了MSI Eureka后, 我们在实验和计算机模拟初期通过使用高质量的严格评估数据, 得以合理化研究计划。MSI Eureka实实在在地使我们的任务更加简单, 我们能更有效地达成研究目标, 并将材料构成实例带入我们的教学工作中。

*Prof. H.J. Seifert,*  
德国弗赖贝格工业大学 (TU Bergakademie Freiberg)



# 客户举例



卡尔斯鲁厄理工学院，既是德国的一所大学同时也是国家级的  
大型研究中心，欧洲第四所理工大学



德国于利希研究中心，德国亥姆霍兹国家研究中心联合会的下属科研机构



伊斯坦布尔科技大学，位于土耳其，成立于1773年



TU Clausthal

德国克劳斯塔尔工业大学：建立于1775年，是德国著名的工业大学之一



巴巴原子能研究中心，印度领先的多学科原子能研究中心，覆盖核科学和工程及相关领域

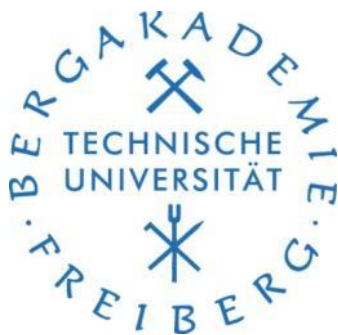
# 客户举例



俄罗斯科学院集团采购  
： 约50家相关研究所



德国亚琛工业大学  
： 德国最富盛名、  
最大的理工科学



德国弗赖贝  
格工业大学



德国德累斯顿工业大学  
： 德国最古老最有声誉  
的大学之一



科威特政府石油公司



洛斯诺普。格鲁门公司。世界第三  
大军工制造商，世界上最大的雷达  
和海军船只制造商

# 提纲

## 一、MSI公司及产品介绍

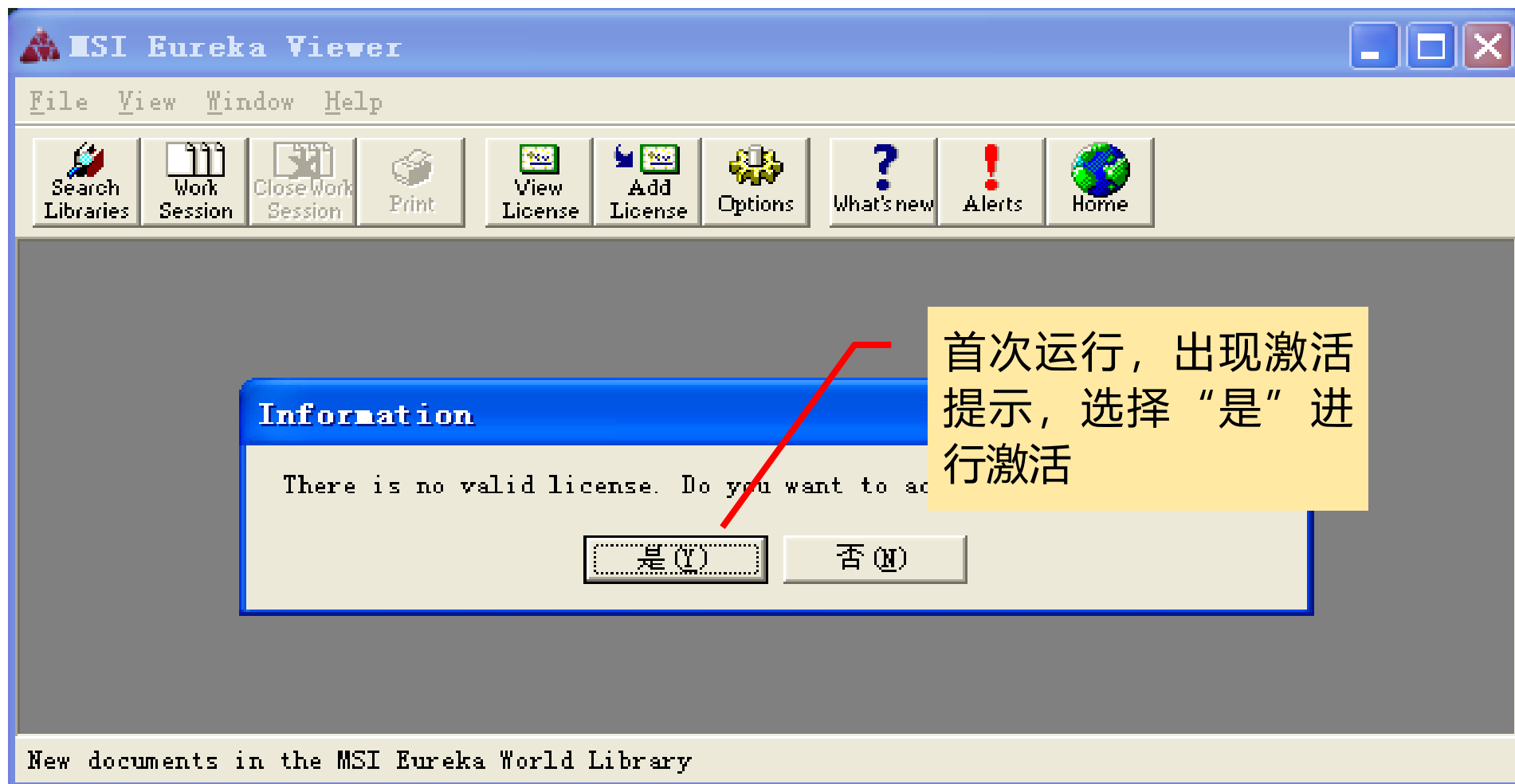
## 二、MSI数据库客户端使用介绍

## 下载&安装

- ▶ 客户端: MSI Eureka Viewer(约7MB)
- ▶ 下载地址:  
<http://www.msiport.com/discover-msi-eureka/how-to-access-msi-eureka/eureka-viewer/download-installation/>
- ▶ 双击进行安装, 请从程序或桌面图标运行该软件

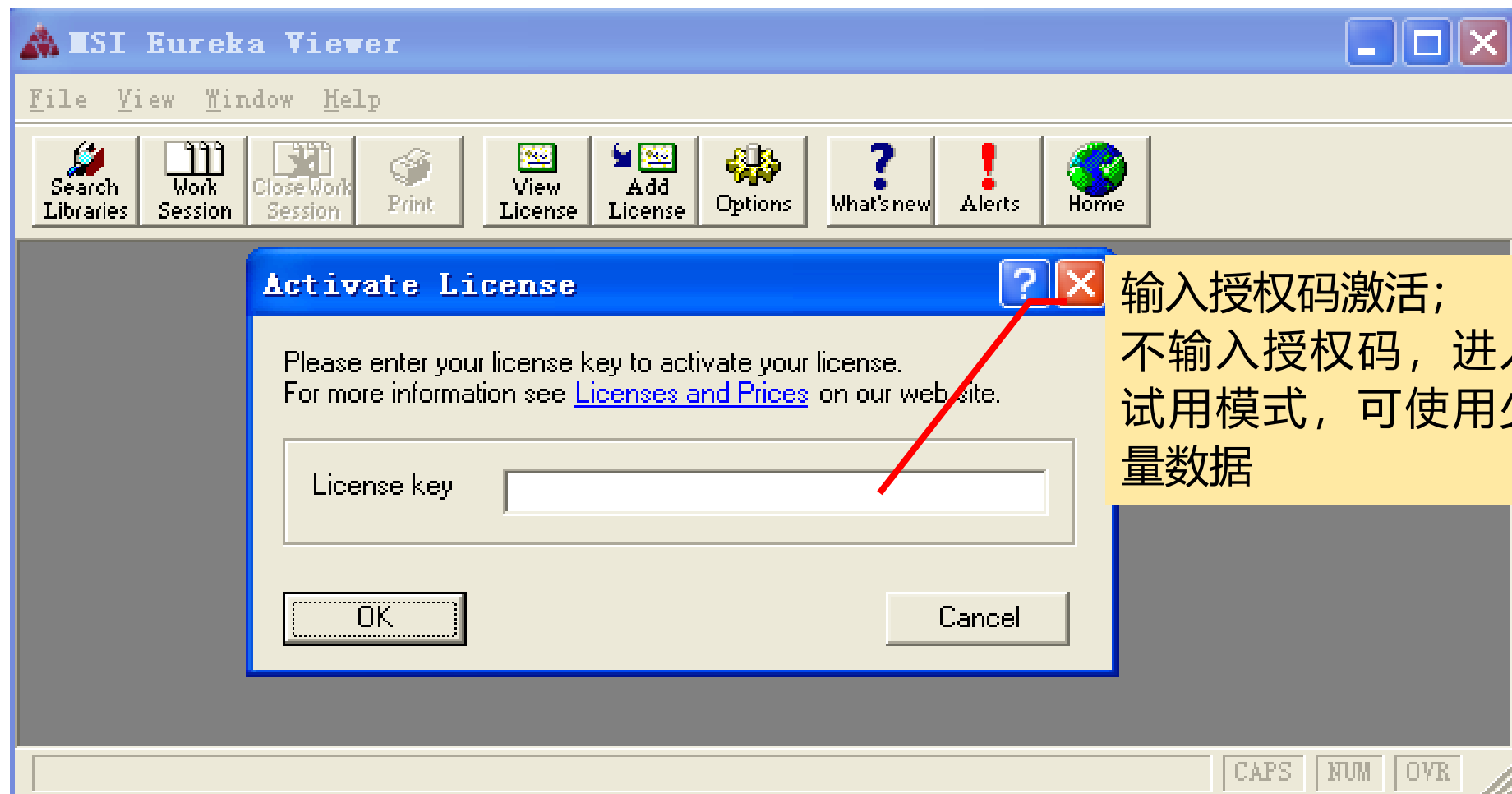


## 激活





## 激活



# 界面简介

**主菜单**

**工具栏**

**MSI网页链接**

**Work Session Contents**

- Ternary Evaluations
  - System Report Al-Cu-Mg
    - Text
      - System Report
    - Tables
      - Table 1: Solid Phases
      - Table 2: Invariant Equilibria
      - Table 3: Reported data for the invariant reaction E2
      - Table 4: Reported data for the invariant reaction E4
      - Table 5: Reported data for the invariant reaction U
      - Table 6: Reported data for the invariant reaction E6
      - Table 7: Lattice parameters of (Al) phase [51Poo]
    - References
      - Literature
    - Reaction Schemes
      - Fig. 6: The reaction scheme, part 1
      - Fig. 6a: The reaction scheme, part 2
    - Diagrams
      - Fig. 2: Liquidus projection with the invariant liquid c
      - Fig. 3: Isothermal section at 400°C from 33.3 to 100
      - Fig. 4: Isothermal section at 400°C [52Ura]
      - Fig. 5: Liquidus projection with liquidus isotherms, c
      - Fig. 7: Isothermal sections in the Al-rich corner at 4
      - Fig. 8: Isothermal sections in Al-rich corner at temp
      - Fig. 9: Solubility of Al and Mg in Cu [57Rog]
    - Schematic
      - Fig. 1: Phases detected along the 33.3 at.% Mg se
- Research Results
- Links to Literature
- Binary Evaluations
- Diagrams as published
- p-T-x Diagrams
- Property Links
- Books

**License information**

**License Information**

Named Account ID: 138-098-084-089-000-061  
 Public Access Point ID: 081-225-233-235-008-068

**Valid Licenses**

Total number: 1

?	Evaluation License	Licensee	Contract/Schedule	Validity	Documents	Categories	Element Systems
D	MSI Eureka Evaluation	D/8	01.01.2012 to 01.01.2013				
1							Ag,Cu,Sn
1							As,Ga,Pt
1							Al,Cu,Mg
1							Al,Mg,Si
1							Fe,Ni,Ti
2							Al,Cu
2							Cu,Ti
3							Au,Sn
3							C,Ti
3							Co,Cr,Fe
3							Cr,Mo,Ni
4							H,Ti
4							Cu,S
5							Au,Te
5							Bi,Pd
5							Ri,Se

# 使用举例——体系选择: Cu-Al-Mg

MSI Eureka Viewer

File View Window Help

Search Libraries Work Session Close Work Session Print View License Add License Options What's new Alerts

Search Libraries

MSI Eureka World Library Contents

Ternary Evaluations	3885	View
Research Results	4295	View
Links to Literature	334587	View
Binary Evaluations	156	View
Diagrams as published	1605	View
p-T-x Diagrams	188	View
Property Links		View
Total.....	344716	

The present MSI Eureka World Library was updated on 15.11.2012

search by combination of

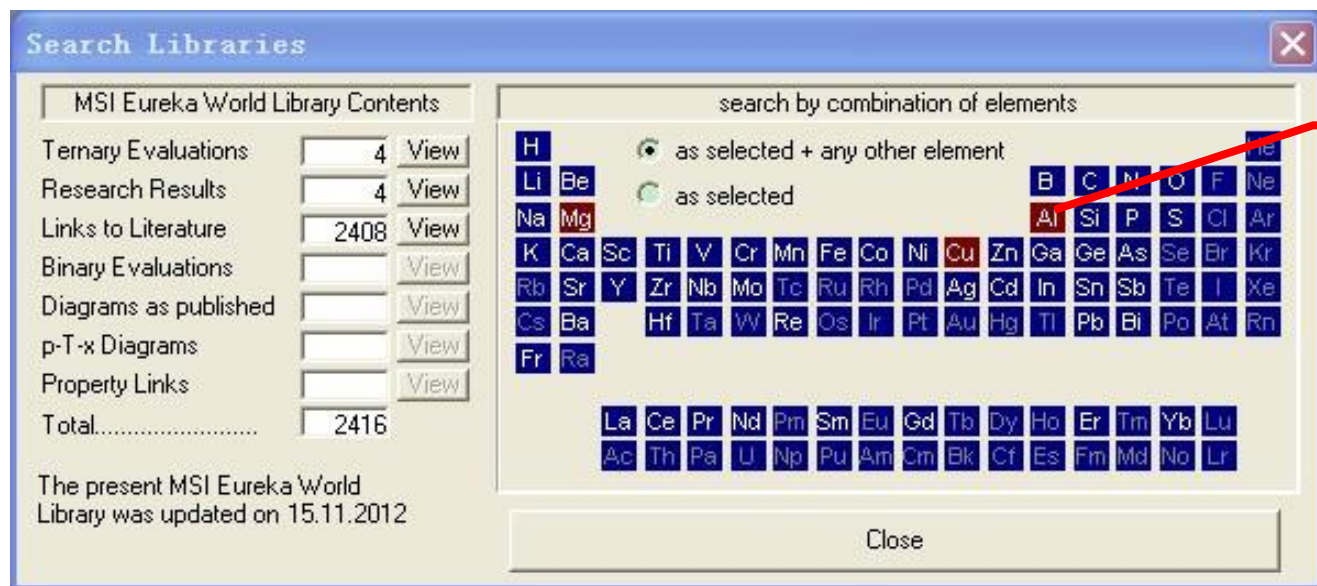
☒ as selected + any other element

☐ as selected

Close

两种选择:  
1.多元: 可选择多种元素;  
2.单元: 只能选择一种元素

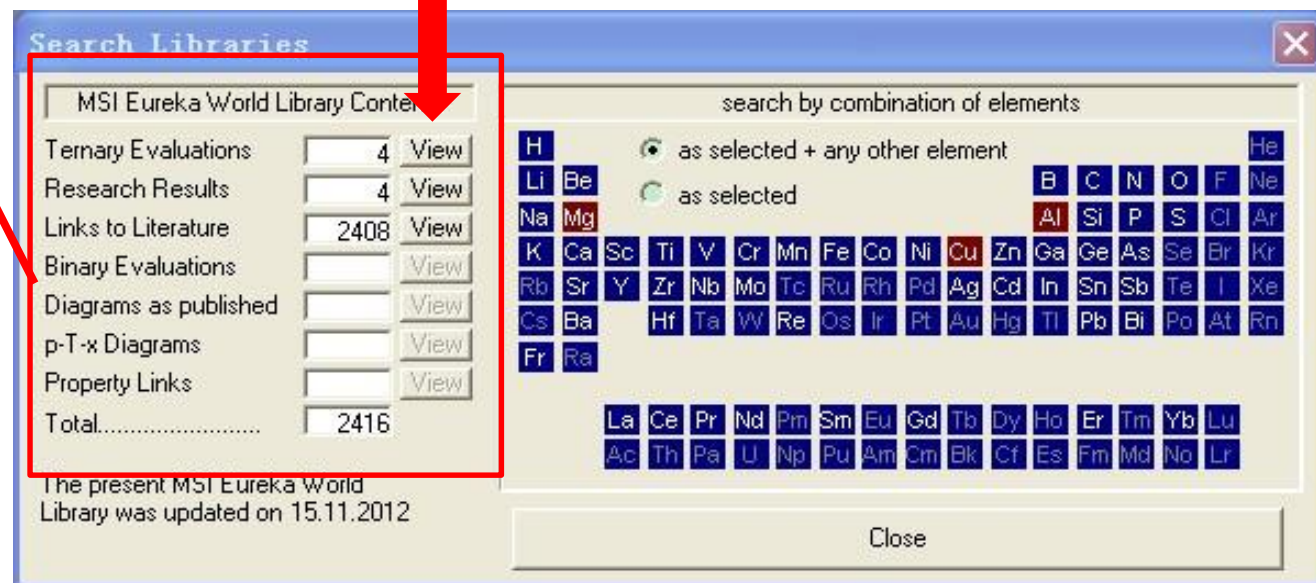
# 使用举例 —— 体系选择: Cu-Al-Mg



☆ 提示：选定的元素背景变棕色，不能选择的元素背景变灰

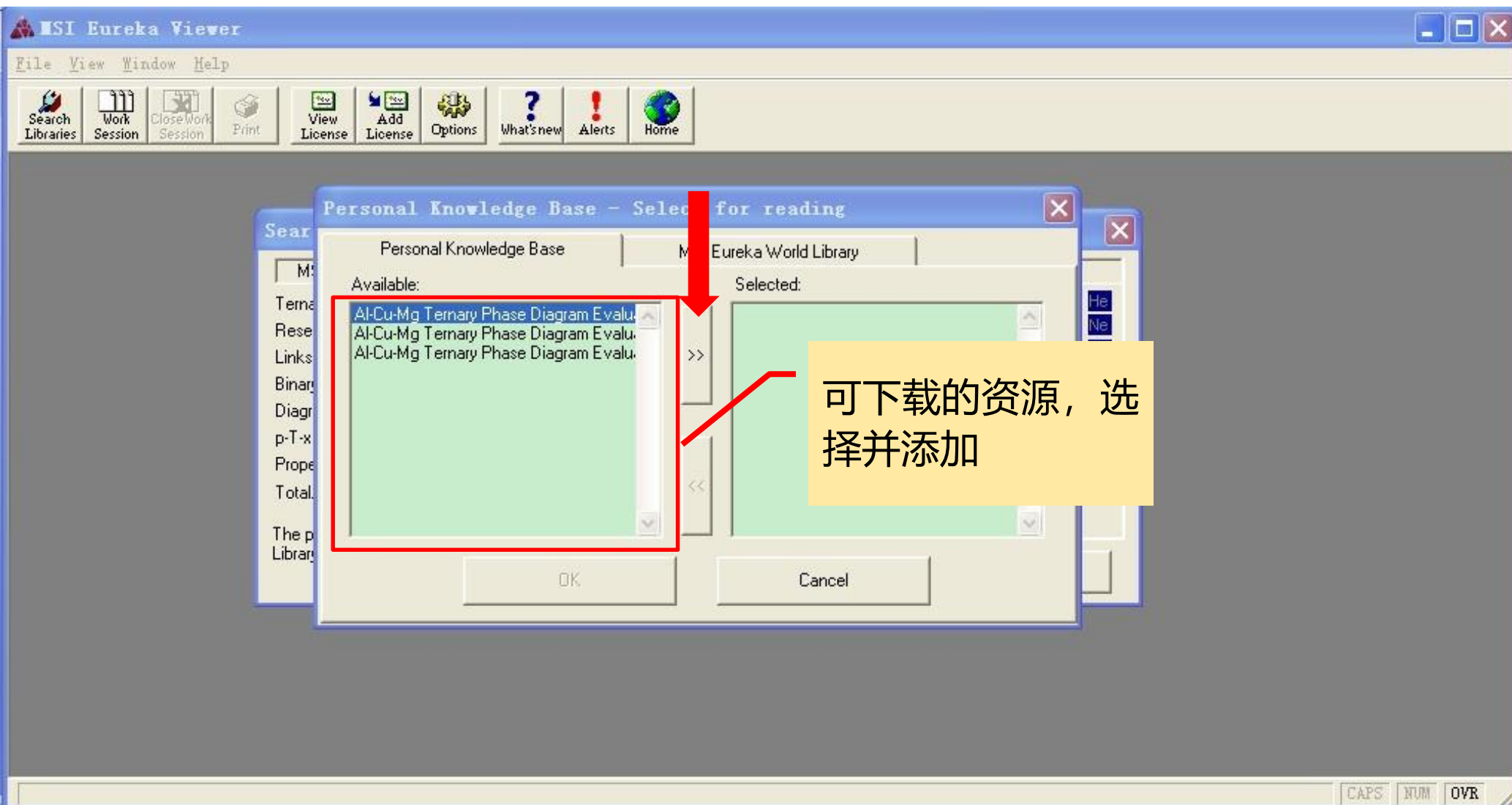
## 使用举例 —— 体系选择: Cu-Al-Mg

数据库所含的内容信息，点击“View”可查看

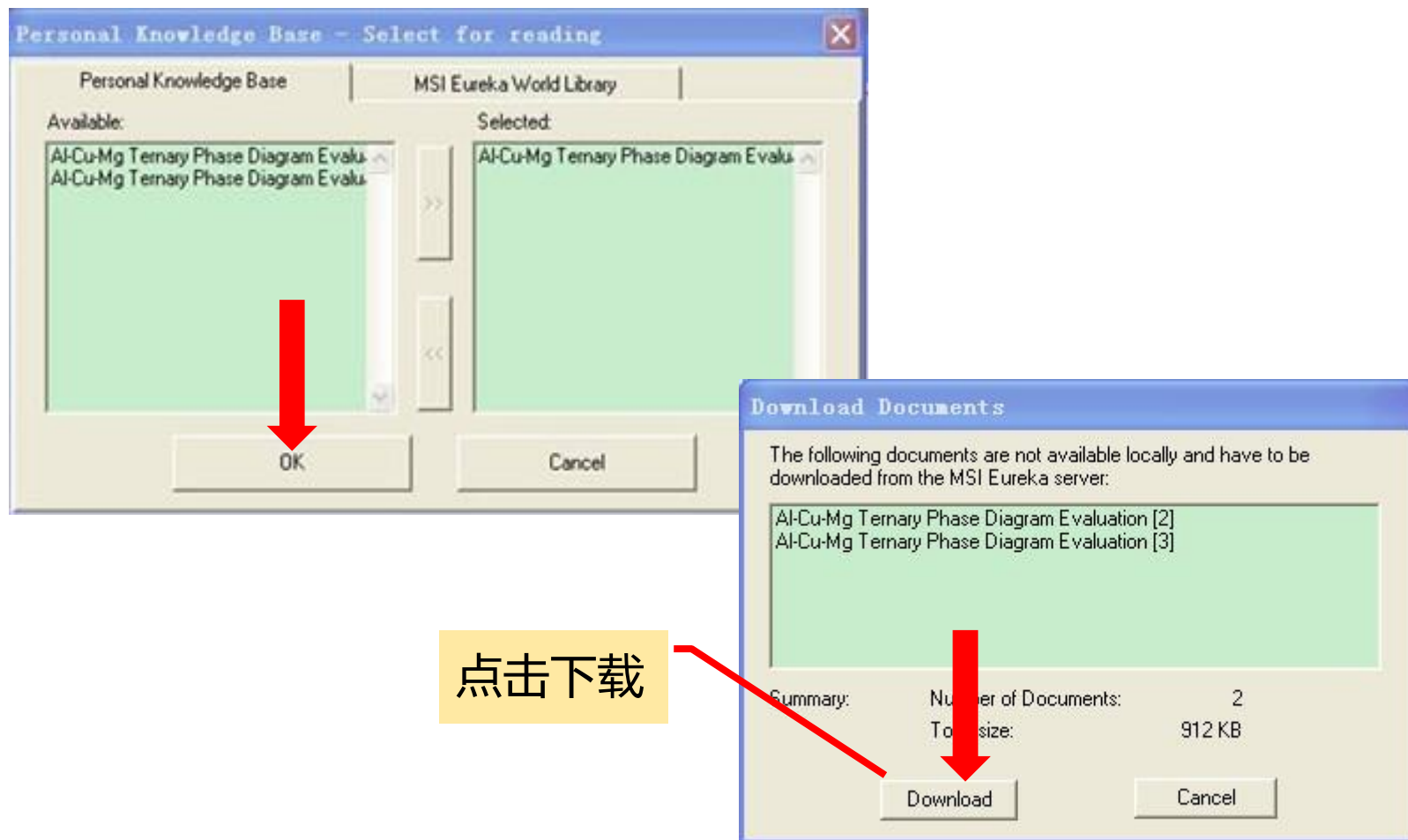




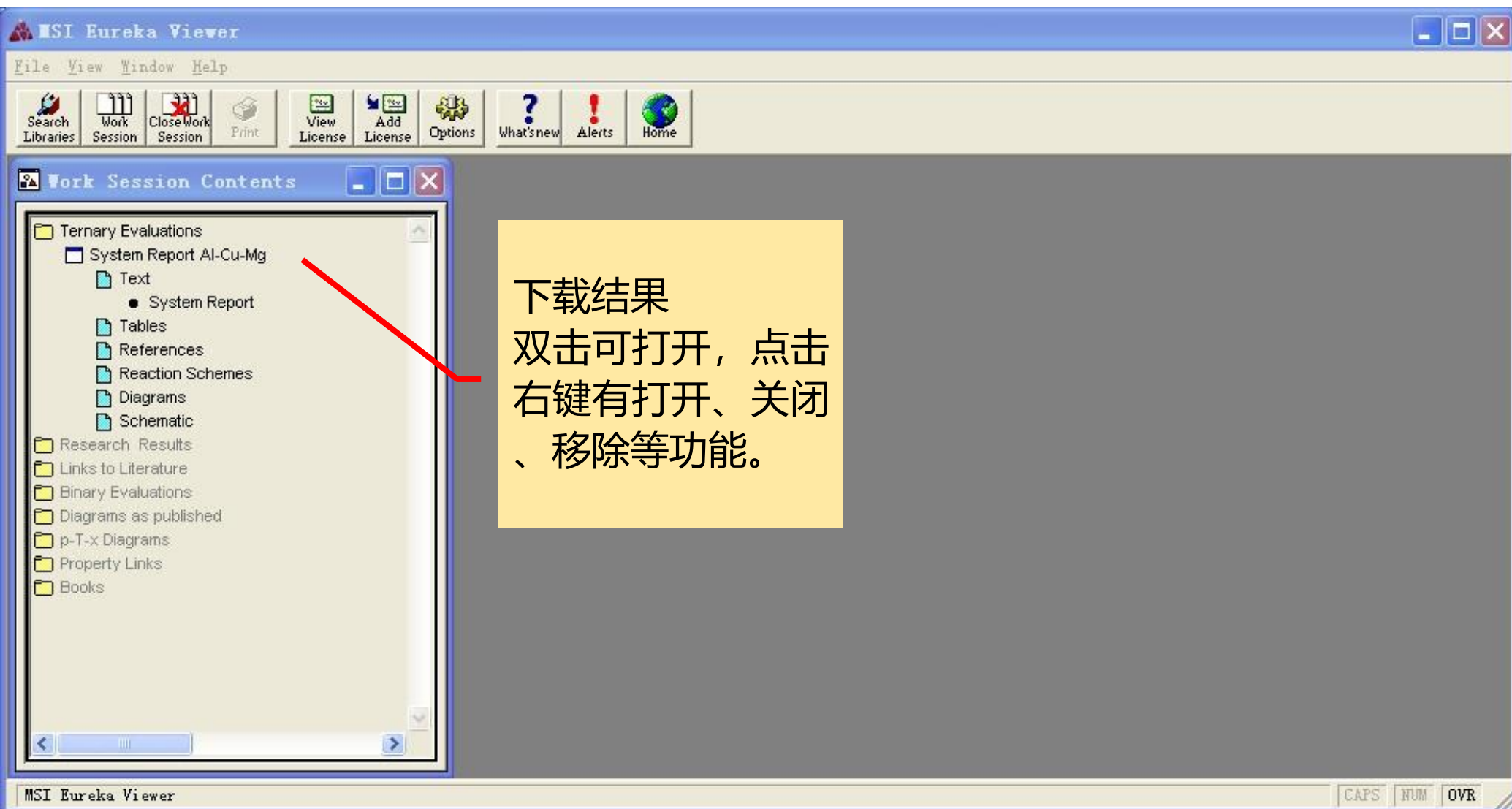
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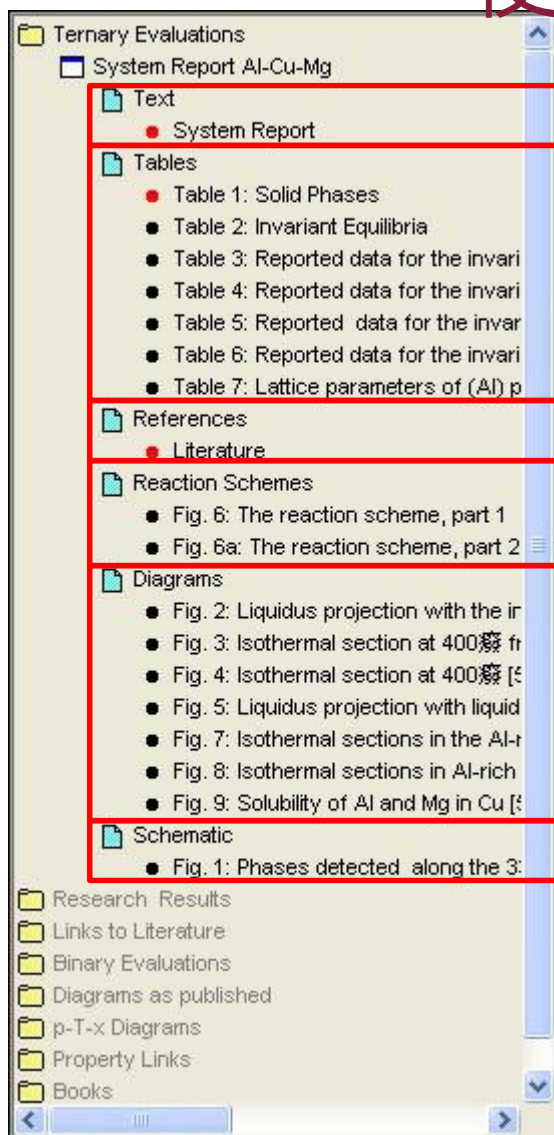
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# 使用举例 —— 数据内容介绍



关于Cu-Al-Mg三元相图的介绍文档，包括二元边界系统，三元固相、液相面，等温截面，零变点等信息

表格信息，包含Cu-Al-Mg三元系中固相、反应、零变点等的温度、成分及其它信息

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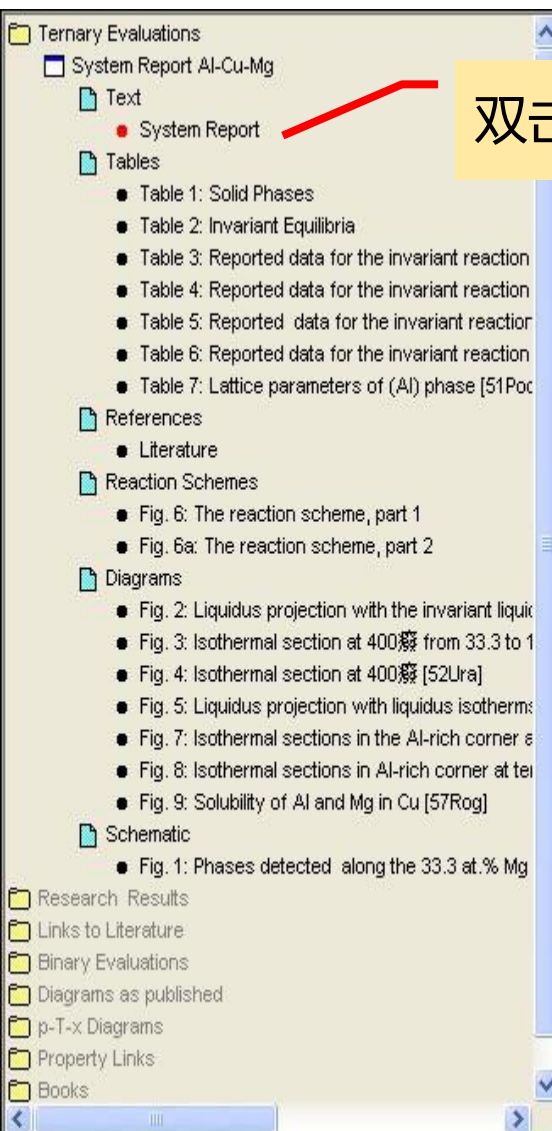
相变过程中的反应流程图

三元截面相图

伪二元系相变过程



# 使用举例 —— System Report



双击打开文档

## Aluminium - Copper - Magnesium

Göster Effenberg and Alan Prince

### Introduction

The equilibria in the Al-Cu-Mg system are complex and not fully understood. The complexity arises from the occurrence of six ternary phases, and the lack of understanding from a need to clarify experimentally the ternary equilibria involving three Laves phases,  $\lambda_{1-3}$ . Four ternary compounds have been studied extensively. The S phase is based on the  $\text{Al}_2\text{CuMg}$  composition, V on  $\text{Al}_{11}\text{Cu}_{11}\text{Mg}_4$  and Q on  $\text{Al}_7\text{Cu}_3\text{Mg}_6$ . Each phase exists over a very limited homogeneity range. The T phase has a broad range of homogeneity and the formula  $(\text{Al}_x\text{Cu}_{1-x})_{49}\text{Mg}_{32}$  adequately describes the composition. Three Laves phases have been identified. The  $\lambda_1$  phase with a  $\text{Cu}_2\text{Mg}$  type structure is a solution phase of the binary  $\text{Cu}_2\text{Mg}$  compound with replacement of the Cu atoms by Al along the 33.3 at.% Mg section. At a composition close to the  $\text{Al}_2\text{Cu}_4\text{Mg}_3$  formula, the  $\lambda_1$  phase melts congruently at  $\approx 940^\circ\text{C}$ . Further replacement of Cu by Al stabilizes the  $\lambda_2$  phase with a  $\text{MgNi}_2$  type structure and then the  $\lambda_3$  phase with a  $\text{MgZn}_2$  type structure. A variety of polytype structures with different atom layer stacking sequences have been observed between the  $\text{MgNi}_2$  and  $\text{MgZn}_2$  type phases. The  $\lambda_{2-3}$  phases appear to be formed by peritectic reaction and each Laves phase is associated with a region in which it forms as the primary phase on solidification of melts. A  $400^\circ\text{C}$  isothermal section, Fig. 3, indicates the equilibrium relationships involving phases  $\lambda_1$ ,  $\lambda_2$  and  $\lambda_3$  for alloys containing 33.3 at.% Mg or more [81Mel2]. No comparable information is available for alloys with less than 33.3 at.% Mg. The liquidus projection, presented by [52Ura], does not include the monovariant curves associated with the  $\text{L} + \lambda_1 = \lambda_2$  and  $\text{L} + \lambda_2 = \lambda_3$  peritectic reactions (Fig. 5). They have been tentatively inserted in the projection of Fig. 2. The Laves phases are the predominant primary phases, but also the regions for primary solidification of (Al) and (Mg) are relatively large. Six pseudobinary reactions have been identified, Table 2, and the pseudobinary reaction e, suggested. The invariant reactions associated with the primary (Al)



region at 400°C to include the AlCuMg phase appears to be justified on the basis of the phase associated with the pseudobinary of being  $\lambda_3$ . [38Pet1] regarded the Cu

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ive in Fig. 4 and does not of [81Mel2]. The Laves with a greater probability proved this assumption.

### Invariant Equilibria

Table 2 lists the invariant reactions proposed for the Al-Cu-Mg ternary system and may be read in conjunction with Fig. 2. The reaction scheme, Fig. 6 and Fig. 6a, is simplified, as  $\beta_0$  and  $\beta$ ,  $\gamma_0$  and  $\gamma$ ,  $\epsilon_1$  and  $\epsilon_2$  and the different Laves phases are not distinguished and denoted by  $\beta$ ,  $\gamma$ ,  $\epsilon$  and  $\lambda$ , respectively. The solid state reactions of the Cu-rich Al-Cu phases and of the Al-Mg phases are neglected. The three phase equilibria with solid phases not stable at room temperature are underlined by dashed lines. The ternary eutectic reaction  $E_2$  has been widely studied, Table 3. The flat nature of the liquidus surface near to  $E_4$  has led to a considerable scatter in quoted compositions and temperatures, Table 4. The reaction has normally been quoted as a ternary eutectic reaction and this is accepted. The transition reaction  $U_{10}$  has also been widely studied, Table 5. The work of [46Ura, 49Ura2] and [48Bro] rests on an examination of a greater number of alloys than other work and allowed a more precise determination of the transition at  $U_{10}$ . Ternary eutectic reactions in Mg-rich alloys occur at  $E_3$  and  $E_6$ . The [34Por] or 2 K [49Ura2] below the binary Cu-Mg eutectic temperature liquid composition at  $E_3$  since they indicate an addition of 0.5% Al compared with 2% Al found by [49Ura2] for a 2 K depression. The involving a Laves phase, but the work of [51Mir2] indicates that this by the previous workers. The liquid composition quoted by [33Bas] composition to be in a primary Laves phase field.

[46Ura]

G.G. URAZOV and D.A. PETROV, "Investigation of the Al-Cu-Mg Phase Diagram" (in Russian), Zhur. Fiz. Khim., 20, 387-398 (1946) (Equi. Diagram, Experimental, 10)

at  $U_{10}$ . [3Bas], for the 1 K regarded as not detected and this

The reactions at  $E_5$ ,  $U_9$ ,  $U_{13}$ ,  $U_{14}$  and  $P_2$  are from Soviet studies. The ternary eutectic reaction at  $E_5$  is due to [49Ura2]. The Al-Mg binary system [81Sch] indicates the peritectic formation of two binary phases designated X and Y. The ternary transition reactions  $U_{11}$  and  $U_{12}$  are introduced as speculative; no experimental data exists to substantiate either reaction. The reaction at  $P_2$  was initially regarded as a transition reaction [37Nis, 52Han], whereas [46Ura] and [49Ura2] considered it to be a ternary peritectic reaction,  $L + \lambda_1 + S \rightleftharpoons T$ . [51Mir2] gives it as  $L + \lambda_1 + S \rightleftharpoons Q$ . There is doubt



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### Invariant Equilibria

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**Table 2: Invariant Equilibria**

T (°C) Reaction	Type	Phase	Composition (at. %)		
			Al	Cu	Mg
665 $L + \varepsilon_2 + \lambda_{1-3} \rightleftharpoons V$	$P_1$	L	45.9	42.0	12.1
520 $L + \lambda_{1-3} + S \rightleftharpoons Q$	$P_2$	L	53.6	6.7	39.7
964-1022 $l + \beta_0 \rightleftharpoons \beta + \gamma_0$	$U_1$				
818 $L + \gamma_0 \rightleftharpoons \lambda_1 + \beta$	$U_2$	L	25.5	58.7	15.8
775 $L + \gamma_0 \rightleftharpoons \varepsilon_1 + \lambda$	$U_3$	L	34.3	50.8	14.9
724 $L + \varepsilon_1 \rightleftharpoons \varepsilon_2 + \lambda_{1-3}$	$U_4$	L	39.5	46.2	14.3
595	$U_5$	L	53.5	38.1	8.4

三元系中零变点处的所有反应信息

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type phase  $\lambda_1$  or to a composition AlCuMg. They take no account of the  $\lambda_2$  and  $\lambda_3$  Laves phases, but extend the  $\lambda_1$  phase region at 400°C to include the AlCuMg composition. This extension is shown as a dashed curve in Fig. 4 and does not appear to be justified on the basis of reports on crystal structure (Fig. 1) and on the later work of [81Mel2]. The Laves phase associated with the pseudobinary reaction at  $p_5$  is more likely to be the  $\lambda_2$  or  $\lambda_3$  phase, with a greater probability of being  $\lambda_3$ . [38Pet1] regarded the CuAl<sub>2</sub>-S section as a pseudobinary, but later work has disproved this assumption.

### Invariant Equilibria

Table 2 lists the invariant reactions proposed for the Al-Cu-Mg ternary system and may be read in conjunction with Fig. 2. The reaction scheme, Fig. 6 and Fig. 6a, is simplified, as  $\beta_0$  and  $\beta$ ,  $\gamma_0$  and  $\gamma$ ,  $\epsilon_1$  and  $\epsilon_2$  and the different Laves phases are not distinguished and denoted by  $\beta$ ,  $\gamma$ ,  $\epsilon$  and  $\lambda$ , respectively. The solid state reactions of the Cu-rich Al-Cu phases and of the Al-Mg phases are neglected. The three phase equilibria with solid phases not stable at room temperature are underlined by dashed lines. The ternary eutectic reaction  $E_2$  has been widely studied, Table 3. The flat nature of the liquidus surface near to  $E_4$  has led to a considerable scatter in quoted compositions and temperatures, Table 4. The reaction has normally been quoted as a ternary eutectic reaction and this is accepted. The transition reaction  $U_{10}$  has also been widely studied, Table 5. The work of [46Ura, 49Ura2] and [48Bro] rests on an examination of a greater number of alloys than other work and allowed a more precise determination of the liquid composition at  $U_{10}$ .

Ternary eutectic reactions in Mg-rich alloys occur at  $E_3$  and  $E_6$ . The reaction temperature at  $E_3$  is 1 K [32Por, 33Bas, 34Por] or 2 K [49Ura2] below the binary Cu-Mg eutectic temperature. Preference is given to the French data for the liquid composition at  $E_3$  since they indicate an addition of 0.5% Al to lower the binary eutectic temperature by 1 K compared with 2% Al found by [49Ura2] for a 2 K depression. The ternary eutectic  $E_6$ , Table 6, was initially regarded as involving a Laves phase, but the work of [51Mir2] indicates that this eutectic involves the Q phase which was not detected by the previous workers. The liquid composition quoted by [33Bas] and [34Por] is not accepted. [40Han] proved this composition to be in a primary Laves phase field.

The reactions at  $E_5$ ,  $U_9$ ,  $U_{13}$ ,  $U_{14}$  and  $P_2$  are from Soviet studies. The ternary eutectic reaction at  $E_5$  is due to [49Ura2]. The Al-Mg binary system [81Sch] indicates the peritectic formation of two binary phases designated X and Y. The ternary transition reactions  $U_{11}$  and  $U_{12}$  are introduced as speculative; no experimental data exists to substantiate either reaction. The reaction at  $P_2$  was initially regarded as a transition reaction [37Nis, 52Han], whereas [46Ura] and [48Bro] regarded it as a eutectic reaction.



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eutectic sections at Cu contents below the beginning of the primary Q phase region. [43Gue, 49Ura1, 49Ura2] and [51Mir2] are in agreement on the nature of these two sections and the most-detailed study [51Mir2] is accepted, Table 2. The  $\gamma_0$ - $\lambda_1$  section is considered as pseudobinary according to liquidus projections of [49Ura1, 49Ura2] and [52Ura] but there has been no direct study of this section. The investigation [51Mir2] of the region of primary solidification of Q led to the conclusion that the T phase is formed by peritectic reaction with Q at  $p_6$ , Fig. 2.

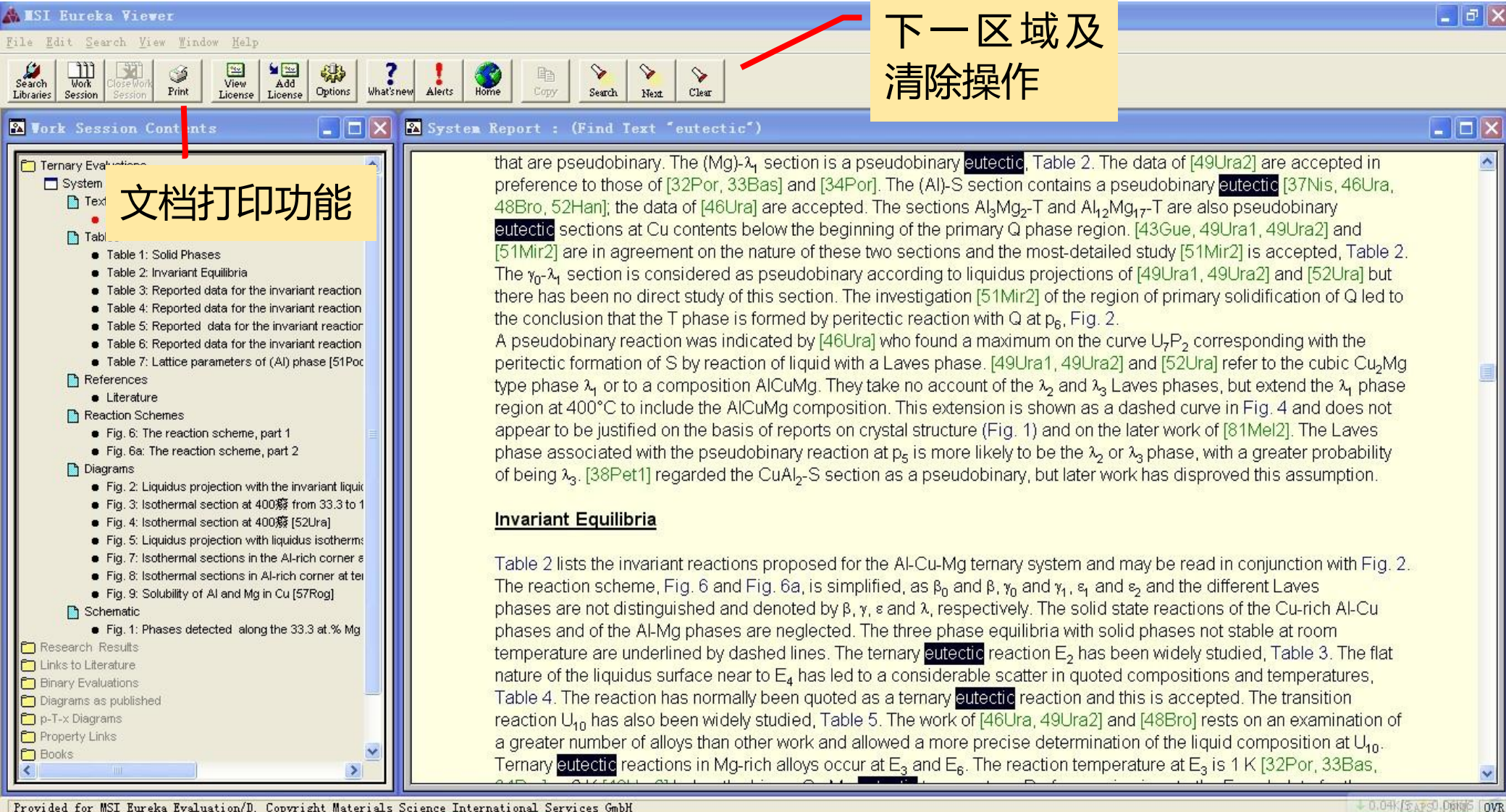
A pseudobinary reaction was indicated by [46Ura] who found a maximum on the curve  $U_7P_2$  corresponding with the peritectic formation of S by reaction of liquid with a Laves phase. [49Ura1, 49Ura2] and [52Ura] refer to the cubic  $Cu_2Mg$  type phase  $\lambda_1$  or to a composition AlCuMg. They take no account of the  $\lambda_2$  and  $\lambda_3$  Laves phases, but extend the  $\lambda_1$  phase region at 400°C to include the AlCuMg composition. This extension is shown as a dashed curve in Fig. 4 and does not appear to be justified. The structure (Fig. 1) and on the later work of [81Mel2]. The Laves phase associated with the  $\lambda_1$  phase is more likely to be the  $\lambda_2$  or  $\lambda_3$  phase, with a greater probability of being  $\lambda_3$ .

**Invariant E**

Table 2 lists the invariant reactions in the Cu-Mg ternary system and may be read in conjunction with Fig. 2. The reaction scheme, Fig. 6 and Fig. 6a, is simplified, as  $\beta_0$  and  $\beta$ ,  $\gamma_0$  and  $\gamma_1$ ,  $\epsilon_1$  and  $\epsilon_2$  and the different Laves phases are not distinguished and denoted by  $\beta$ ,  $\gamma$ ,  $\epsilon$  and  $\lambda$ , respectively. The solid state reactions of the Cu-rich Al-Cu phases and of the Al-Mg phases are neglected. The three phase equilibria with solid phases not stable at room temperature are underlined by dashed lines. The ternary eutectic reaction  $E_2$  has been widely studied, Table 3. The flat nature of the liquidus surface near to  $E_4$  has led to a considerable scatter in quoted compositions and temperatures, Table 4. The reaction has normally been quoted as a ternary eutectic reaction and this is accepted. The transition reaction  $U_{10}$  has also been widely studied, Table 5. The work of [46Ura, 49Ura2] and [48Bro] rests on an examination of a greater number of alloys than other work and allowed a more precise determination of the liquid composition at  $U_{10}$ . Ternary eutectic reactions in Mg-rich alloys occur at  $E_3$  and  $E_6$ . The reaction temperature at  $E_3$  is 1 K [32Por, 33Bas, 34Por] or 2 K [49Ura2] below the binary Cu-Mg eutectic temperature. Preference is given to the French data for the liquid composition at  $E_3$  since they indicate an addition of 0.5% Al to lower the binary eutectic temperature by 1 K compared with 2% Al found by [49Ura2] for a 2 K depression. The ternary eutectic  $E_6$ , Table 6, was initially regarded as

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that are pseudobinary. The (Mg)- $\lambda_1$  section is a pseudobinary **eutectic**, Table 2. The data of [49Ura2] are accepted in preference to those of [32Por, 33Bas] and [34Por]. The (Al)-S section contains a pseudobinary **eutectic** [37Nis, 46Ura, 48Bro, 52Han]; the data of [46Ura] are accepted. The sections  $\text{Al}_3\text{Mg}_2$ -T and  $\text{Al}_2\text{Mg}_{17}$ -T are also pseudobinary **eutectic** sections at Cu contents below the beginning of the primary Q phase region. [43Gue, 49Ura1, 49Ura2] and [51Mir2] are in agreement on the nature of these two sections and the most-detailed study [51Mir2] is accepted, Table 2. The  $\gamma_0$ - $\lambda_1$  section is considered as pseudobinary according to liquidus projections of [49Ura1, 49Ura2] and [52Ura] but there has been no direct study of this section. The investigation [51Mir2] of the region of primary solidification of Q led to the conclusion that the T phase is formed by peritectic reaction with Q at  $p_6$ , Fig. 2.

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**Invariant Equilibria**

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Fig. 2: Liquidus projection with the invariant liquid compositions ...

data curves & grid: atom% of element  
axes scaling: atom% of element

Al

(Al) E<sub>2</sub> E<sub>6</sub> E<sub>7</sub> E<sub>8</sub>

Mg

Cu

Fig. 2: Liquidus projection with the invariant liquid compositions labelled to correspond with Table 2

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[43Gue, 49Ura1, 49Ura2] and study [51Mir2] is accepted, Table 2. [49Ura1, 49Ura2] and [52Ura] but of primary solidification of Q led to

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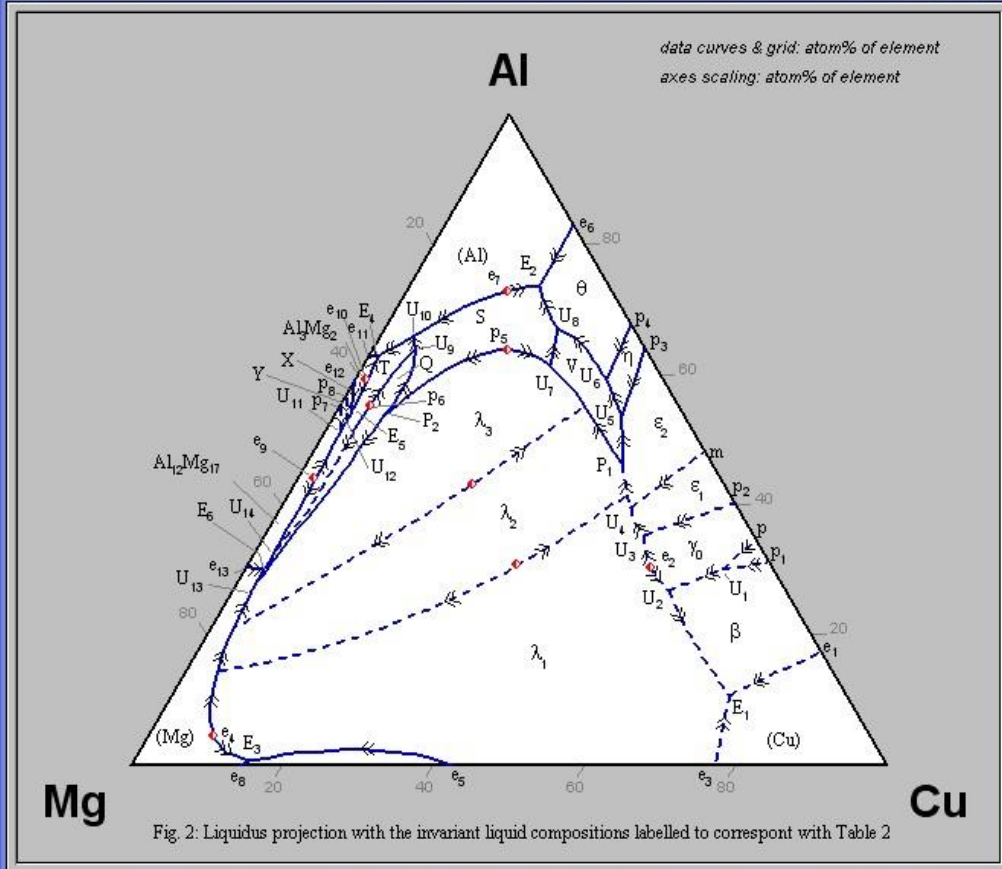


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- ☐ System Report Al-Cu-Mg
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  - Tables
    - Table 1: Solid Phases
    - Table 2: Invariant Equilibria

Diagram Attributes

Atom% Mass%

Mg	<div style="width: 24.2%;"></div>	24.2
Cu	<div style="width: 54.6%;"></div>	54.6
Al	<div style="width: 42.8%;"></div>	42.8

- Fig. 5: Liquidus projection with liquidus isotherms
- Fig. 7: Isothermal sections in the Al-rich corner
- Fig. 8: Isothermal sections in Al-rich corner at temperature
- Fig. 9: Solubility of Al and Mg in Cu [57Rog]
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Fig. 2: Liquidus projection with the invariant liquid compositions ...

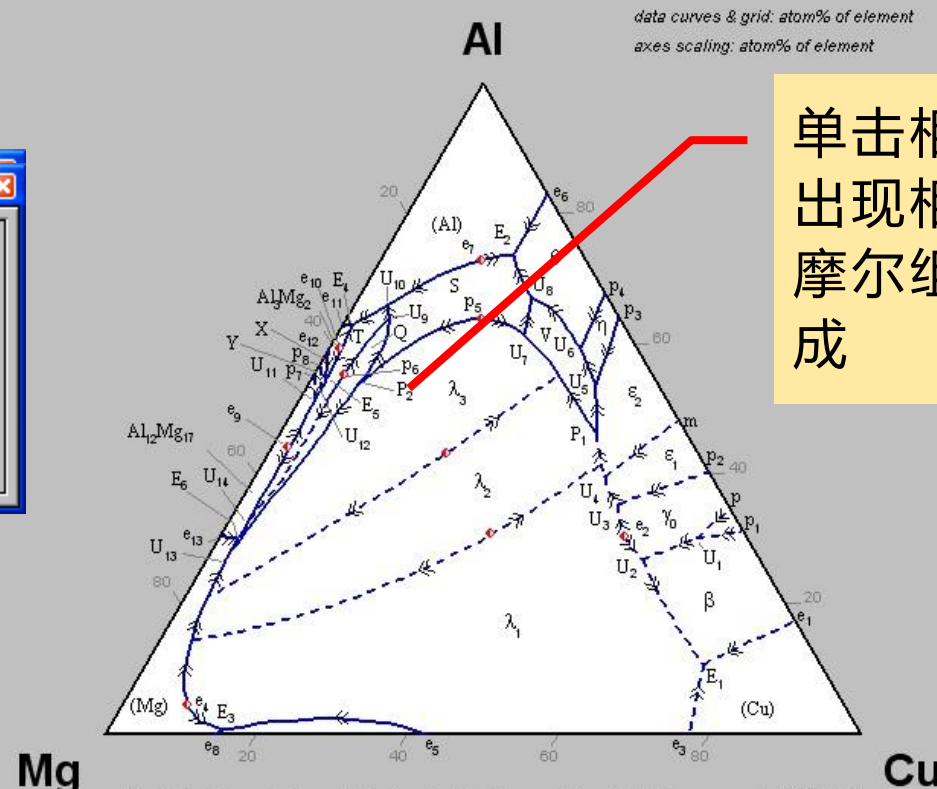


Fig. 2: Liquidus projection with the invariant liquid compositions labelled to correspond with Table 2

单击相图中某个点，表格中出现相应的组成，绿色代表摩尔组成，红色代表质量组成

43Gue, 49Ura1, 49Ura2] and study [51Mir2] is accepted, Table 2. [49Ura1, 49Ura2] and [52Ura] but

may be read in conjunction with Fig. 2. and the different Laves reactions of the Cu-rich Al-Cu phases not stable at room temperature have been widely studied, Table 3. The flat compositions and temperatures, is accepted. The transition [48Bro] rests on an examination of of the liquid composition at U<sub>10</sub>. temperature at E<sub>3</sub> is 1 K [32Por, 33Bas, given to the French data for the eutectic temperature by 1 K, Table 6, was initially regarded as

放大功能，单击出现空白区域

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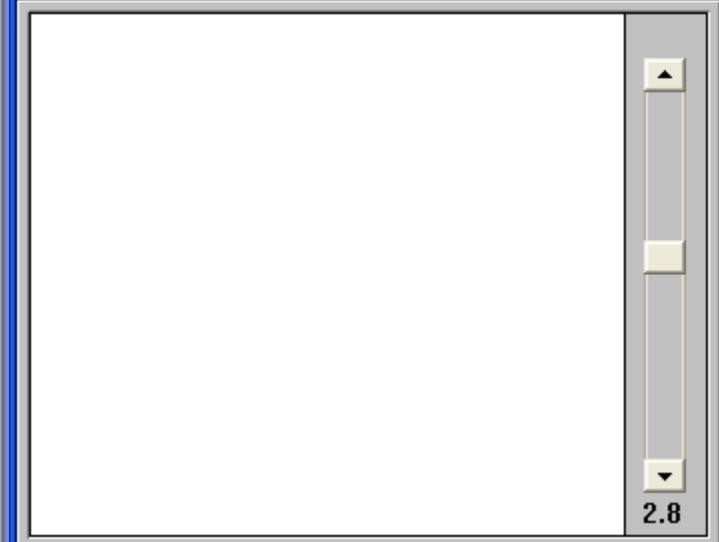


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Magnifying Glass



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Fig. 2: Liquidus projection with the invariant liquid compositions ...

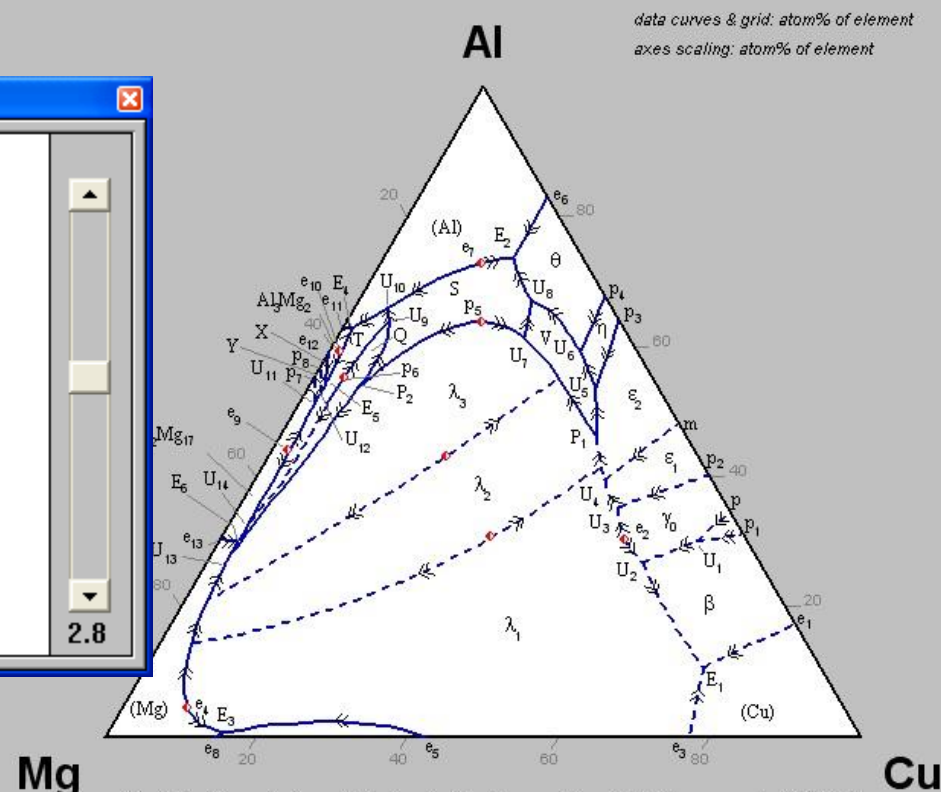


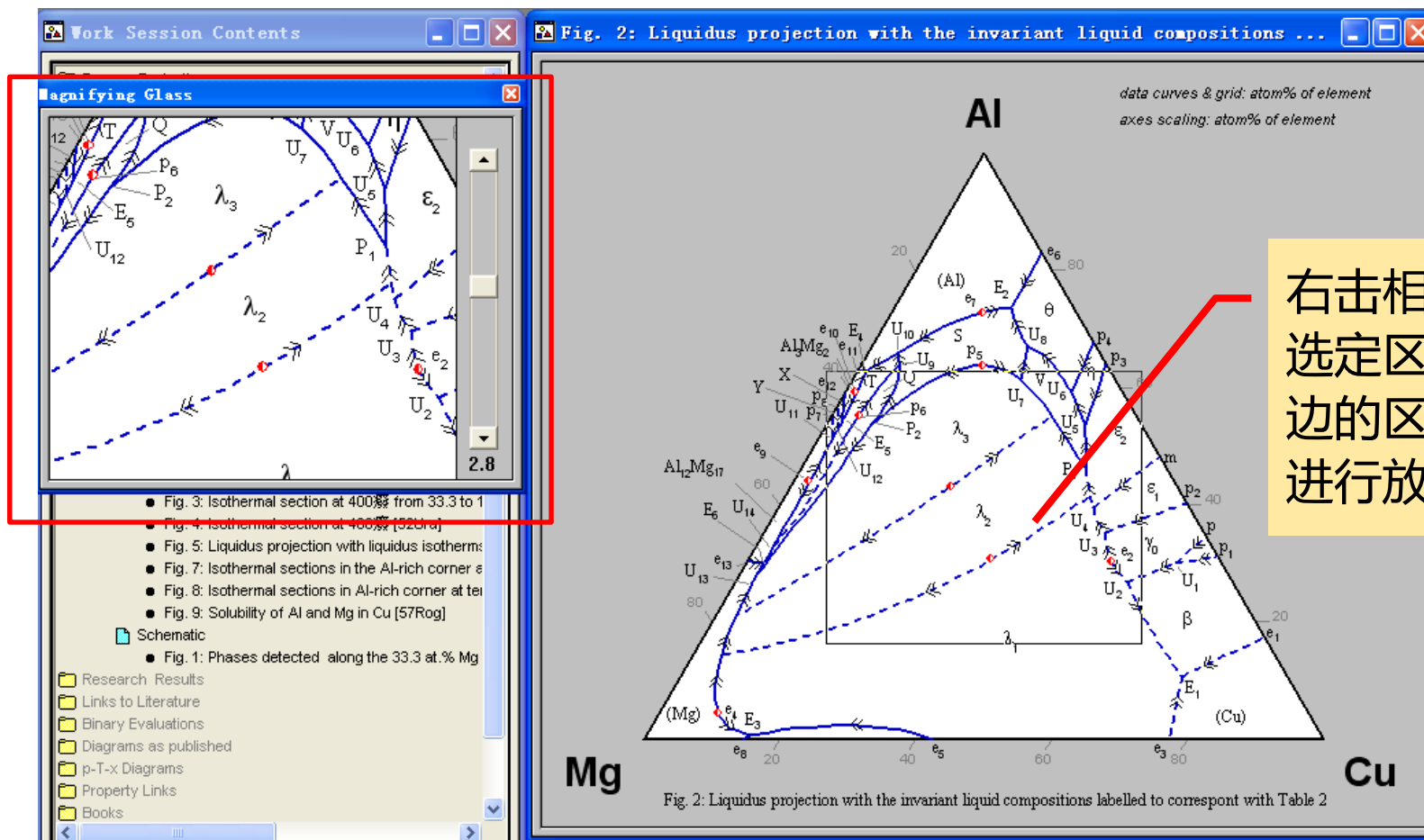
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the  $U_7P_2$  corresponding with the and [52Ura] refer to the cubic  $Cu_2Mg$  phases, but extend the  $\lambda_1$  phase shed curve in Fig. 4 and does not enter work of [81Mel2]. The Laves  $\beta_3$  phase, with a greater probability has disproved this assumption.

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右击相图，系统自动  
选定区域并出现在左  
边的区域中，可对其  
进行放大或缩小操作



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Fig. 2: Liquidus projection with the invariant liquid compositions ...

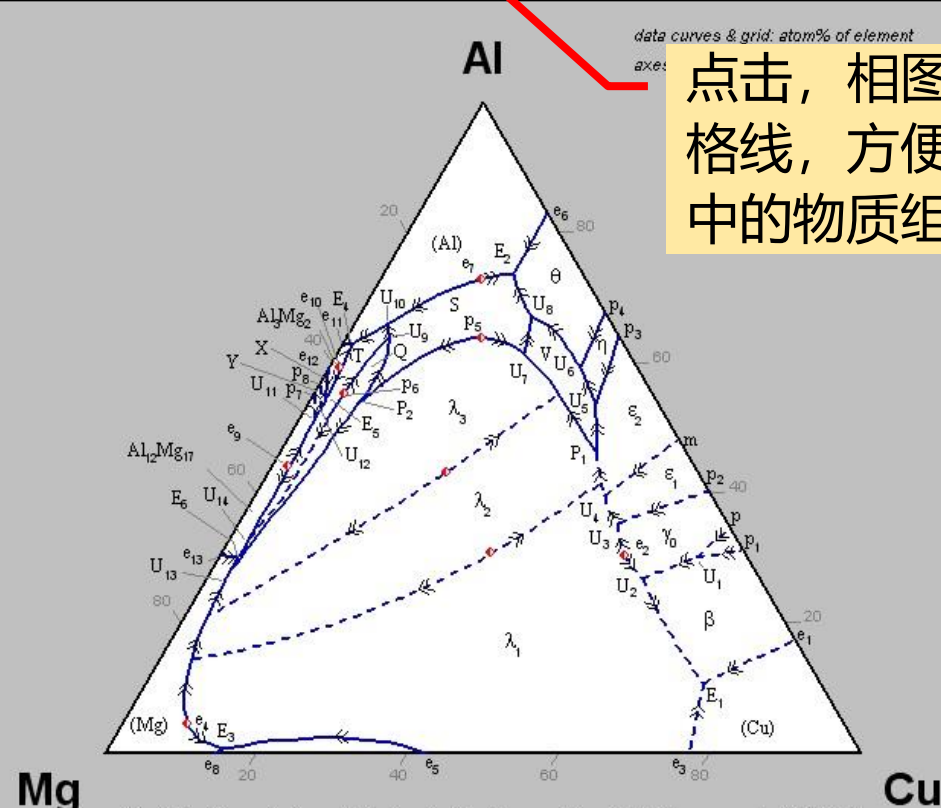


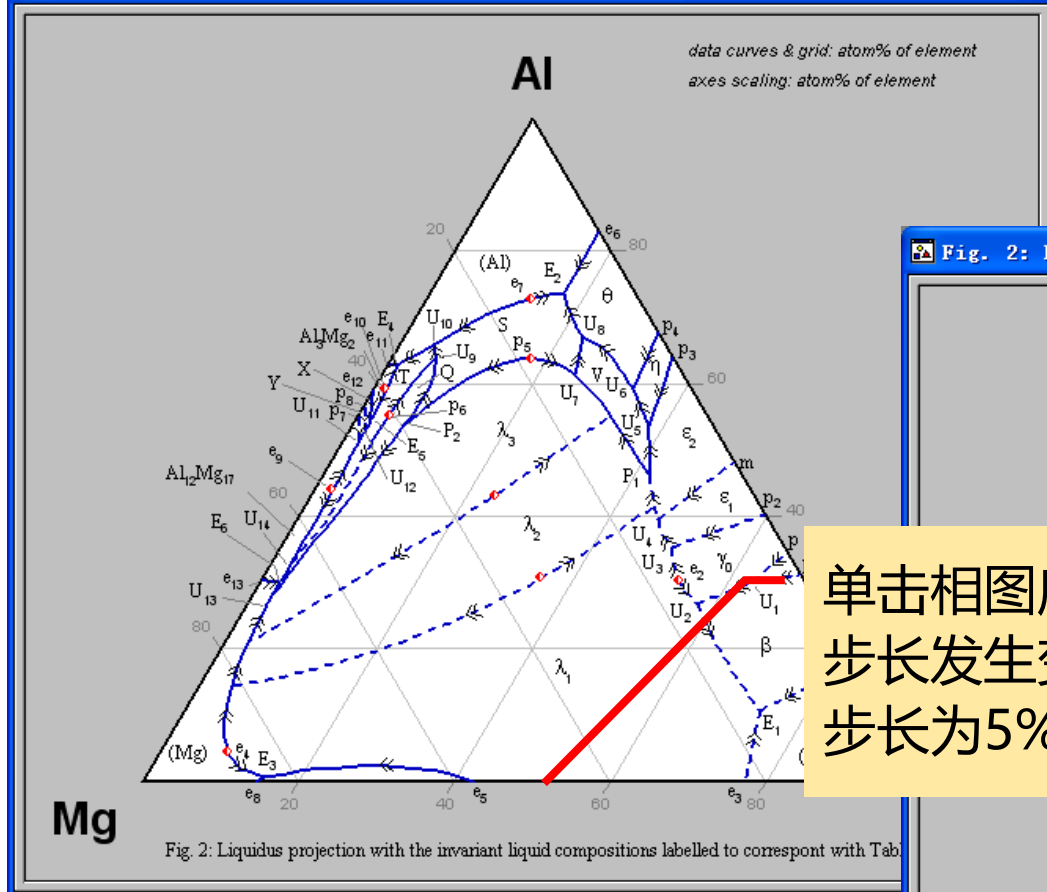
Fig. 2: Liquidus projection with the invariant liquid compositions labelled to correspond with Table 2

点击，相图中出现网格线，方便读取相图中的物质组成

43Gue, 49Ura1, 49Ura2] and [54Ura] is accepted, Table 2. and [52Ura] but the location of Q led to ing with the the cubic Cu<sub>2</sub>Mg and the λ<sub>1</sub> phase shed curve in Fig. 4 and does not ter work of [81Mel2]. The Laves β<sub>3</sub> phase, with a greater probability has disproved this assumption.

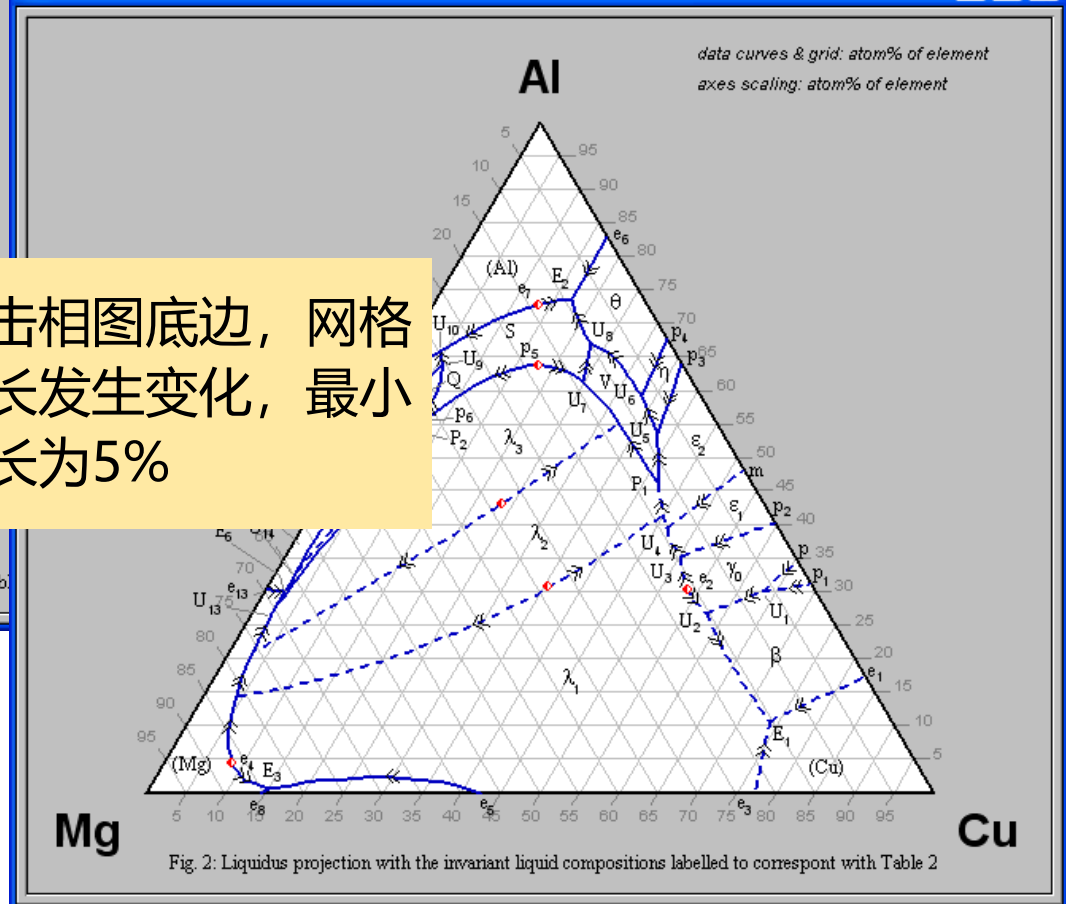
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Fig. 2: Liquidus projection with the invariant liquid compositions ...

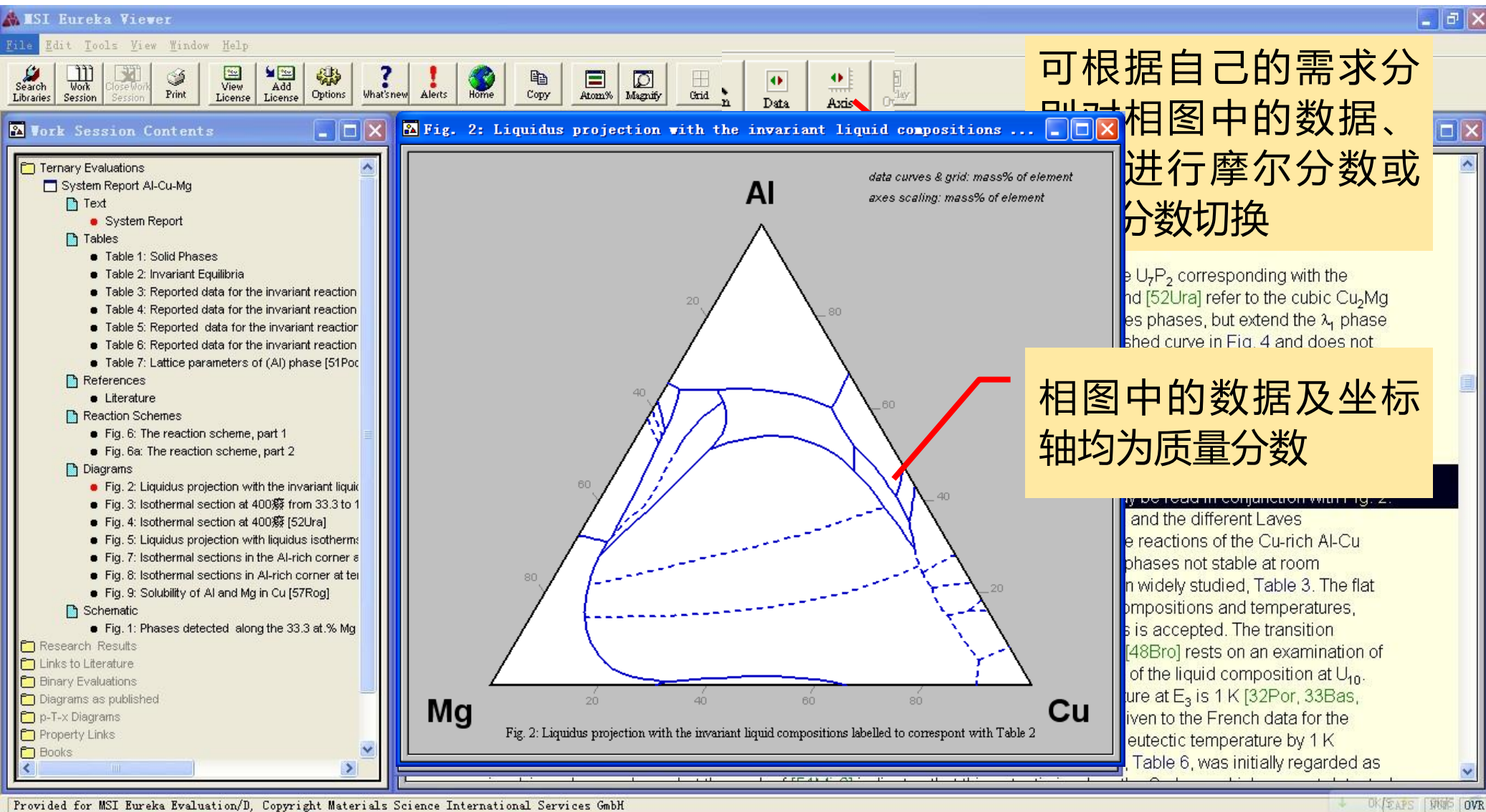


单击相图底边，网格  
步长发生变化，最小  
步长为5%

Fig. 2: Liquidus projection with the invariant liquid compositions ...



# 使用举例 —— 相图查阅





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重叠功能，可对两个  
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Fig. 6a: The reaction scheme, part 2

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Fig. 8: Isothermal sections in Al-rich corner at

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Fig. 1: Phases detected along the 33.3 at.% Mg

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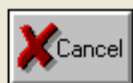
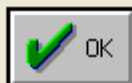
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Books

Overlay

Compatible Diagrams :

Fig. 2: Liquidus projection with the



选定需要比较的图形  
， 点击 “OK”

Fig. 2: Liquidus projection

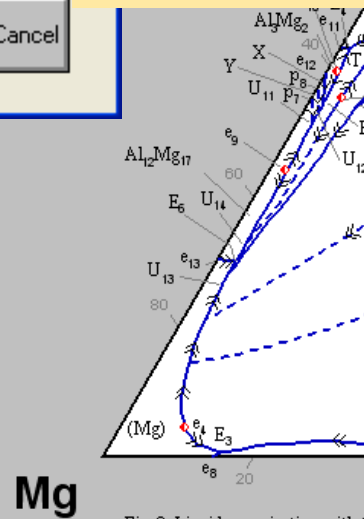


Fig. 2: Liquidus projection with the

Fig. 3: Isothermal section at 400°C from 33.3 to 100 at.% Mg [81Me12]

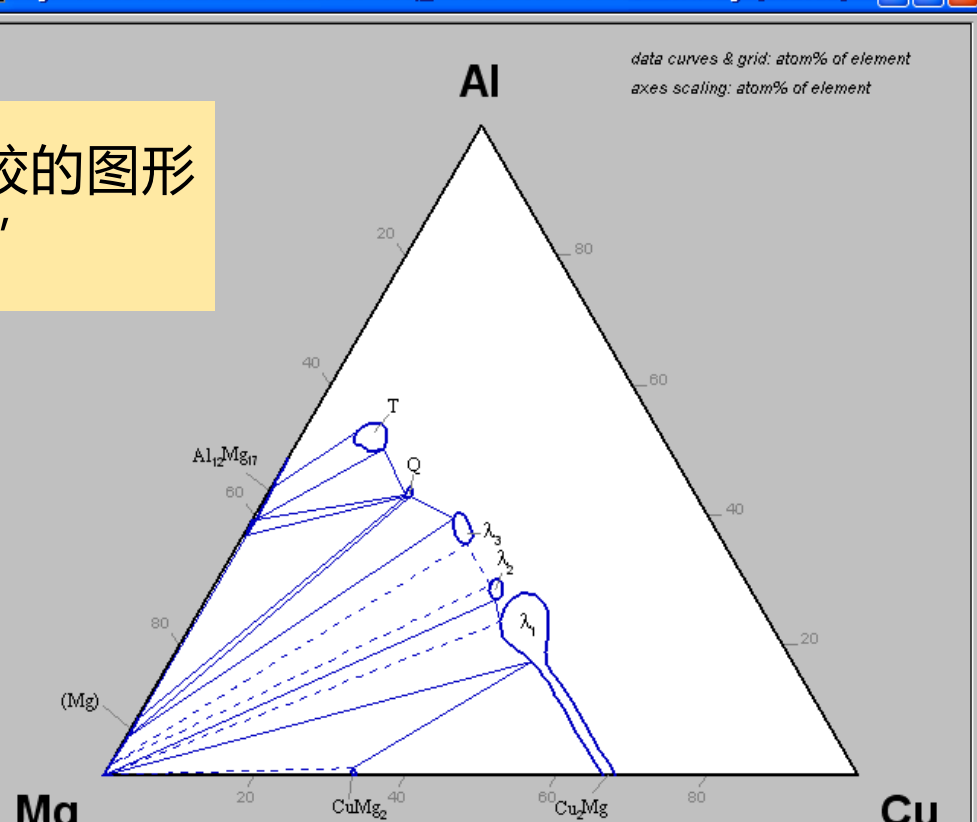
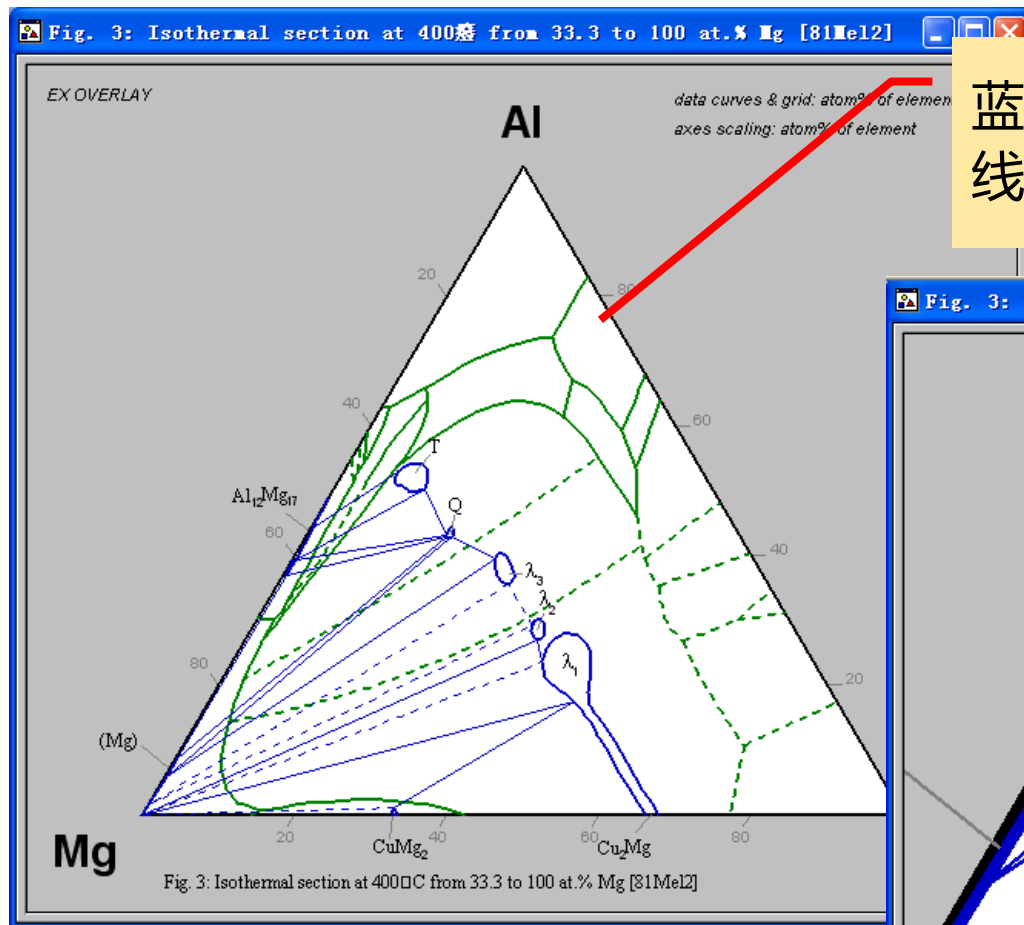


Fig. 3: Isothermal section at 400°C from 33.3 to 100 at.% Mg [81Me12]

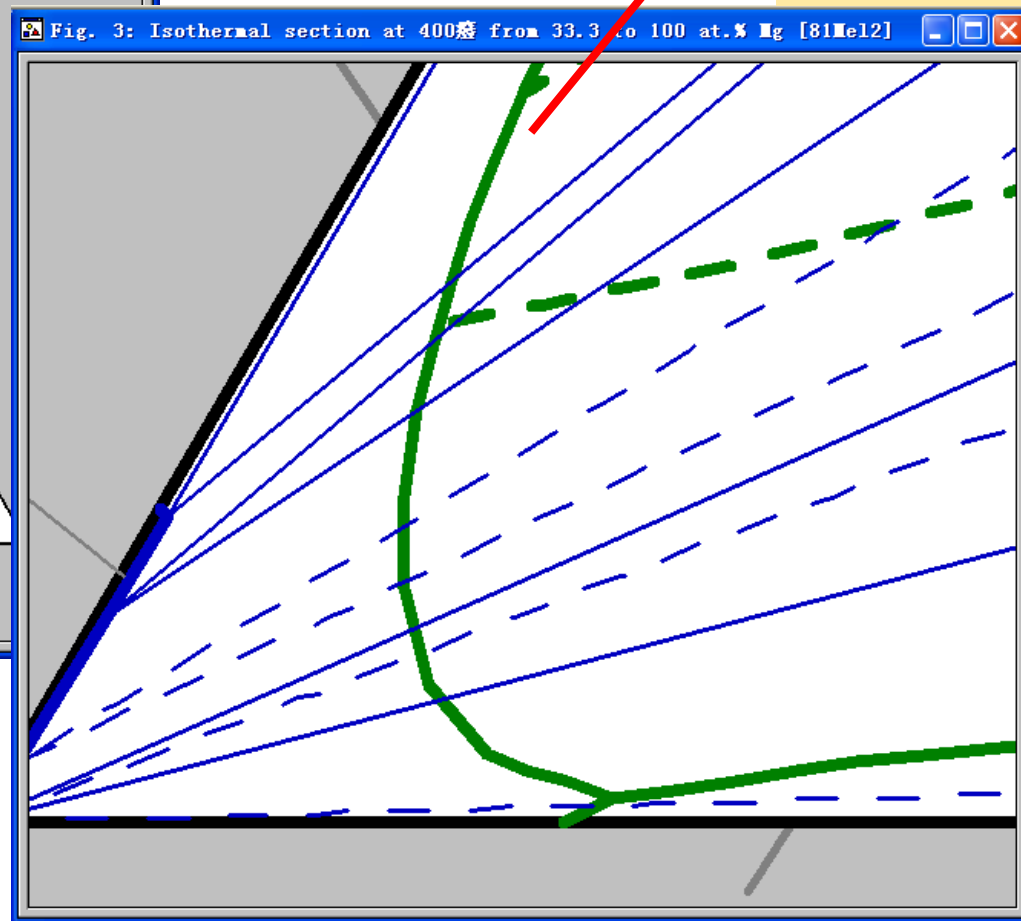
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0.1K/2APS 100% OVR



蓝色线为原图，绿色  
线为叠加上去的相图

放大相图的  
部分区域



# MSI

Science Simplified



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